On the relevance of generalized disclinations in defect mechanics

Chiqun Zhang and Amit Acharya

Carnegie Mellon University, Pittsburgh, PA 15213, USA

Abstract

The utility of the notion of generalized disclinations in materials science is discussed within the physical context of modeling interfacial and bulk line defects like defected grain and phase boundaries, dislocations and disclinations. The Burgers vector of a disclination dipole in linear elasticity is derived, clearly demonstrating the equivalence of its stress field to that of an edge dislocation. We also prove that the inverse deformation/displacement jump of a defect line is independent of the cut-surface when its g.disclination strength vanishes. An explicit formula for the displacement jump of a single localized composite defect line in terms of given g.disclination and dislocation strengths is deduced based on the Weingarten theorem for g.disclination theory (Weingarten-gd theorem) at finite deformation. The Burgers vector of a g.disclination dipole at finite deformation is also derived.

1 Introduction

While the mechanics of disclinations has been studied [DeW69, DeW71, DeW73a, DeW73b, Nab87, HPL06, Zub97, RK09, FTC11], there appears to be a significant barrier to the adoption of disclination concepts in the practical modeling of physical problems in the mechanics of materials, perhaps due to the strong similarities between the fields of a disclination dipole and a dislocation [RK09]. Furthermore, the finite deformation version of disclination theory is mathematically intricate [Zub97, DZ11], and does not lend itself in a natural way to the definition of the strength of a disclination purely in terms of any candidate field that may be defined to be a disclination density. This has prevented the introduction of a useful notion of a disclination fields and its computational implementation to generate approximate solutions for addressing practical problems in the mechanics of materials and materials science.

A recent development in this regard is the development of g.disclination theory (generalized disclination theory) [AF12, AF15], that alleviates the significant road-block in the finite deformation setting mentioned above. It does so by adopting a different conceptual standpoint in defining the notion of g.disclinations than what arose in the works of Weingarten and Volterra (as described by Nabarro [Nab87]). This new standpoint also allows the consideration of phase and grain-boundaries and their terminating line defects within a common framework. Briefly, Weingarten asked a question adapted to the theory of linear elasticity which requires the construction of a displacement field on a multiply-connected¹ body with a single hole, and the characterization of its jumps across any surface that renders the body simply-connected when 'cut' by it. An important constraint of the construction is that the strain of the displacement match a given symmetric second-order tensor field on the simply-connected body induced by the cut; the given symmetric second-order tensor field

¹We refer to any non simply-connected body as multiply-connected or multi-connected.

is assumed twice-differentiable on the original multiply-connected domain and to satisfy the St.-Venant compatibility conditions. There is a well defined analogous question at finite deformations [Cas04]. The constructed displacement field on the simply-connected domain will in general have a jump (i.e. difference) in the values of its rotation field at corresponding points across the surface but the jump in its strain (similarly defined) necessarily vanishes by definition. However, when viewed from this perspective and keeping physically abundant objects like (incoherent) phase boundaries in mind across which strains are discontinuous as well, there seems to be no reason to begin from a starting point involving a continuous strain field; it is just as reasonable to ask that one is given a smooth third-order tensor field that is curl-free on a multiply-connected domain with a hole (this condition replacing the given strain field satisfying the St.-Venant compatibility condition), and then ask for the construction of a displacement field whose second gradient matches the given third-order tensor field on a cut-surface induced simply-connected domain, and the characterization of the jump of the displacement field across the surface. This allows the whole first gradient of the deformation/displacement field (constructed on the simply-connected domain) to exhibit jumps across the surface, instead of only the rotation. Moreover, this whole argument goes through seamlessly in the context of geometrically nonlinear kinematics; the g.disclination strength is defined as a standard contour integral of the given third order tensor field. The framework naturally allows the calculation of fields of a purely rotational disclination specified as a g.disclination density distribution.

The principal objectives of this paper are to

- review the physical situations that may be associated with the mathematical concept of disclinations (considered as a special case of g.disclinations). Much is known in this regard amongst specialists (cf. [RK09]), and we hope to provide complementary, and on occasion new, perspective to what is known to set the stage for solving physical problems related to disclination mechanics in a forthcoming paper [ZAP16] within the framework of g.disclinations.
- Establish the connection between the topological properties of a g.disclination dipole and a dislocation at finite strains by deducing the formula for the Burgers vector of the g.disclination dipole.
- Interpret the Weingarten theorem for g.disclinations at finite deformation [AF15] in terms of g.disclination kinematics.

This paper is organized as follows. Section 2 provides a review of related prior literature. A brief Section 3 introduces the notation utilized in the paper. In Section 4 we discuss various physical descriptions for disclinations, dislocations, and grain boundaries as well as the interrelations between them. In Section 5 we derive the Burgers vector of a disclination dipole within classical elasticity theory, making a direct connection with the stress field of an edge dislocation. Section 6 provides an overview of generalized disclination statics from [AF15]. In Section 6.1, the Weingarten theorem for generalized disclination theory from [AF15] is recalled for completeness and a new result proving that the displacement jump is independent of the cut-surface under appropriate special conditions is deduced. In this paper, we refer the Weingarten theorem for generalized disclination theory from [AF15] as the Weingarten-gd theorem. In Section 7 the Weingarten-gd theorem is interpreted in the context of g.disclination kinematics, providing an explicit formula for the displacement jump of a single g.disclination in terms of data prescribed to define the two defect densities (g.disclination and dislocation densities) in g.disclination theory. Finally, in Section 8 we derive the Burgers vector for a g.disclination dipole.

2 A brief review of prior work

In this section we briefly review some of the vast literature on the mathematical modeling of disclinations and grain boundaries. An exhaustive review of the subject is beyond the scope of this paper.

The definition of the dislocation and the disclination in *solids*² was first introduced by Volterra (as described by Nabarro [Nab87]). Nabarro [Nab87] studied geometrical aspects of disclinations and Li [Li72] presented microscopic interpretations of a grain boundary in terms of a dislocation and a disclination model. The static fields of dislocations and disclinations along with applications have been studied extensively within linear elasticity by DeWit [DeW69, DeW71, DeW73b, DeW72], as well as in 2-d nonlinear elasticity by Zubov [Zub97] and the school of study led by Romanov [RK09]. In [RK09], the elastic fields and energies of the disclination are reviewed and the disclination concept is applied to explaining several observed microstructures in crystalline materials. In [RV92], the expression for the Burgers vector for a single-line, two-rotation-axes disclination dipole appropriate for geometrically linear kinematics is motivated from a physical perspective without dealing with questions of invariance of the physical argument w.r.t different cut-surfaces or the topological nature of the displacement jump of a disclination-dipole in contrast to that of a single disclination.

In Fressengeas et al. [FTC11], the elasto-plastic theory of dislocation fields [Ach01] is nontrivially extended to formulate and study time-dependent problems of defect dynamics including both disclination and dislocation fields. Nonlinear elasticity of disclinations and dislocations in 2-d elastic bodies is discussed in [Zub97, DZ11]. Dislocations and disclinations are studied within Riemannian geometry in [KMS15]; Cartan's geometric method to study Riemannian geometry is deployed in [YG12] to determine the nonlinear residual stress for 2-d disclination distributions. Similar ideas are also reviewed in [CMB06], in a different degree of mathematical detail and without any explicit calculations, in an effort to develop a time-dependent model of mesoscale plasticity based on disclination-dislocation concepts. Another interesting recent work along these lines is the one in [RG17] that discusses 'metrical disclinations' (among other things) that are related to our g.disclinations. The concerns of classical, nonlinear disclination theory related to defining the strength of a single disclination in a practically applicable manner, and therefore studying the mechanics of interactions of collections of individual such defects, remains in this theory and the authors promote the viewpoint of avoiding any type of curvature line-defects altogether.

From the materials science perspective, extensive studies have been conducted on grain boundary structure, kinetics, and mechanics from the atomistic [SV83a, SV83b] as well as from more macroscopic points of view [Mul56, CMS06, Cah82]. In [KRR06, SEIDRR04, Roh10, Roh11], the grain boundary character distribution is studied from the point of view of grain boundary microstructure evolution. In [KLT06, EES⁺09], a widely used framework for grain boundary network evolution, which involves the variation of the boundary energy density based on misorientation, is proposed. In most cases, these approaches do not establish an explicit connection with the stress and elastic deformation fields caused by the grain boundary [HHM01]. Phase boundary mechanics considering effects of stress is considered in [PL13, AD15], strictly within the confines of compatible elastic deformations.

One approach to study a low angle grain boundary is to model it as a series of dislocations along the boundary [RS50, SV83a, SV83b]. In [DXS13], a systematic numerical study is conducted of the structure and energy of low angle twist boundaries based on a generalized Peierls-Nabarro model. The dislocation model has also been applied to study grain boundaries with disconnections [HP96, HH98, HPL06, HPL07, HPH09, HP11, HPH⁺13]. In these work, disconnections are modeled

²There is a difference in meaning between disclinations in solids and in nematic liquid crystals as explained in [RK09, KF08, PAD15].

as dislocations at a step and the grain boundaries are represented as a series of coherency dislocations. Long-range stress fields for disconnections in tilt walls are discussed from both the discrete dislocations and the disclination dipole perspectives in [AZH08]. In [VD13, VAKD14, VD15] a combination of the Frank-Bilby equation [Fra50, Fra53, BBS55, SB95, BB56] and anisotropic elasticity theory is employed to formulate a computational method for describing interface dislocations. In [DSSS98] atomic-level mechanisms of dislocation nucleation is examined by dynamic simulations of the growth of misfitting films.

Although low angle grain boundaries can be modeled by dislocations, the dislocation model is no longer satisfactory for describing high-angle boundaries because the larger misorientations require introducing more dislocations along the boundary interface which shortens the distance between dislocations [BAC05]. Thus, it is difficult to identify the Burgers vectors of grain boundary dislocations in high angle grain boundaries. Alternatively, a grain boundary can also be modeled as an array of disclination dipoles [RK09, NSB00]. In [FTC14], the crossover between the atomistic description and the continuous representation is demonstrated for a tilt grain boundary by designing a specific array of disclination dipoles. Unlike the dislocation model for a grain boundary, the disclination model is applicable to the modeling of both low and high angle grain boundaries.

3 Notation and terminology

The condition that a is defined to be b is indicated by the statement a := b. The Einstein summation convention is implied unless otherwise specified. Ab is denoted as the action of a tensor A on a vector b, producing a vector. A \cdot represents the inner product of two vectors; the symbol ADrepresents tensor multiplication of the second-order tensors A and D. A third-order tensor is treated as a linear transformation on vectors to a second-order tensors.

We employ rectangular Cartesian coordinates and components in this paper; all tensor and vector components are written with respect to a rectangular Cartesian basis (e_i) , i=1 to 3. The symbol *div* represents the divergence, *grad* represents the gradient on the body (assumed to be a domain in ambient space). In component form,

$$(\boldsymbol{A} \times \boldsymbol{v})_{im} = e_{mjk} A_{ij} v_k$$

$$(\boldsymbol{B} \times \boldsymbol{v})_{irm} = e_{mjk} B_{irj} v_k$$

$$(\operatorname{div} \boldsymbol{A})_i = A_{ij,j}$$

$$(\operatorname{div} \boldsymbol{B})_{ij} = B_{ijk,k}$$

$$(\operatorname{curl} \boldsymbol{A})_{im} = e_{mjk} A_{ik,j}$$

$$(\operatorname{curl} \boldsymbol{B})_{irm} = e_{mjk} B_{irk,j},$$

where e_{mjk} is a component of the alternating tensor X.

 \mathbf{F}^{e} is the elastic distortion tensor; $\mathbf{W} := (\mathbf{F}^{e})^{-1}$ is the inverse-elastic 1-distortion tensor; \mathbf{S} is the eigenwall tensor (3rd-order); \mathbf{Y} is the inverse-elastic 2-distortion tensor (3rd-order); $\boldsymbol{\alpha}$ is the dislocation density tensor (2nd-order) and $\boldsymbol{\Pi}$ is the generalized disclination density tensor (3rd-order). The physical and mathematical meanings of these symbols will be discussed subsequently in Section 6.

In dealing with questions related to the Weingarten-gd theorem, we will often have to talk about a vector field \boldsymbol{y} which will generically define an inverse elastic deformation from the current deformed configuration of the body.



(a) A compatible/coherent phase boundary, where all atomic planes from either side match along the interface.

(b) An incompatible/incoherent phase boundary. There are some mismatches of the atomic planes along the interfaces.

Figure 1: Illustration of a compatible/coherent and an incompatible/incoherent phase boundary.

4 Basic ideas for the description of g.disclinations, dislocations, and grain boundaries

In this section we will discuss various aspects of modeling g.disclinations and their relationship to dislocations, mostly from a physical perspective and through examples. Beginning from a geometric visualization of single disclinations, we will motivate the physical interpretation of such in lattice structures. The formation and movement of a disclination dipole through a lattice will be motivated. Descriptions of a dislocation in terms of a disclination dipole will be discussed. Finally, we will demonstrate how the description of a low-angle boundary in terms of disclination dipoles may be understood as a dislocated grain-boundary in a qualitative manner.

In many situations in solid mechanics it is necessary to consider a 2-D surface where a distortion measure is discontinuous. In elasticity a distortion corresponds to the deformation gradient; in linear elasticity the distortion will be grad \boldsymbol{u} , where \boldsymbol{u} is the displacement. However, there are many cases where the distortion field cannot be interpreted as a gradient of a vector field on the whole body. In such cases, the distortion will have an incompatible part that is not curl-free. One familiar situation is to consider the presence of dislocations modeled by the elastic theory of dislocations [Krö81, Wil67]. A 2-d surface of discontinuity of the elastic distortion is referred to as a phase boundary, of which the grain boundary is a particular case. Based on whether atomic planes from either side of the interface can match with each other at the interface or not, a phase boundary is categorized into a compatible/coherent or an incompatible/incoherent boundary, as shown in Figure 1. A special compatible phase boundary is called a twin boundary with a highly symmetrical interface, where one crystal is the mirror image of the other, also obtained by a combination of shearing and rotation of one side of the interface with respect to the other. A grain boundary is an interface between two grains with different orientations. The orientation difference between the two grains comprising a grain boundary is called the misorientation at the interface, and it is conventional to categorize grain boundaries based on the misorientation angle. Low angle grain boundaries (LAGBs) are defined as those whose misorientations are less than 11 degrees and high angle grain boundaries (HAGBs) are those with greater misorientations. In the situation that the phase boundary discontinuity shows gradients along the surface, we will consider the presence of



Figure 2: Descriptions of Volterra dislocations and disclinations . Figure (a) is a cylinder with an inner hole along the axis. Figure (b) and (c) are the edge dislocations. Figure (d) is the screw dislocation. Figure (e) and (f) are twist disclinations and Figure (g) is the wedge disclination. (Figure reprinted from [Nab87] with permission from Dover Publications).

line defects. Following [AF12, AF15], the terminating tip-curves of phase boundary discontinuities are called generalized disclinations or g.disclinations.

The classical singular solutions for defect fields contain interesting subtleties. For instance, the normal strain e_{11} in dimension two for a straight dislocation and of a straight disclination in linear elasticity are [DeW71, DeW73b]:

Straight Dislocation

$$e_{11} = -\frac{b_1}{4\pi (1-\nu)} [(1-2\nu) \frac{x_2}{\rho^2} + 2\frac{x_1^2 x_2}{\rho^4}] + \frac{b_2}{4\pi (1-\nu)} [(1-2\nu) \frac{x_1}{\rho^2} - 2\frac{x_1 x_2^2}{\rho^4}],$$

Straight Disclination

$$e_{11} = \frac{\Omega_3}{4\pi(1-\nu)} \left[(1-2\nu)\ln\rho + \frac{x_2^2}{\rho^2} \right].$$

The strain fields blow up in both the dislocation and the disclination cases. In addition, on approaching the core (i.e. the coordinate origin in the above expressions) in dislocation solutions, the elastic strain blows up as $\frac{1}{\rho}$, ρ being the distance from the dislocation core. Thus the linear elastic energy density diverges as $\frac{1}{\rho^2}$, causing unbounded total energy for a finite body for the dislocation whereas the total energy of a disclination is bounded. The disclination, however, has more energy stored in the far-field (w.r.t the core) than the dislocation, and this is believed to be the reason for a single disclination being rarely observed as opposed to a dislocation. Our modeling philosophy and approach enables defects to be represented as non-singular defect lines and surfaces, always with bounded total energy (and even local stress fields).

4.1 Disclinations

Volterra [Nab85] described dislocations and disclinations by considering a cylinder with a small inner hole along the axis, as shown in Figure 2 (the hole is exaggerated in the figure). Figure



Figure 3: A planar illustration for wedge disclinations. Figure (a) is a negative wedge disclination, where a wedge is inserted into a vertical cut causing compressive circumferential stress after 'welding' the wedge to the body. Figure (b) is a positive wedge disclination, where the wedge is taken out of the original structure and the exposed faces welded together. ω is the wedge angle as well as the magnitude of the Frank vector. (Figure reproduced from [Naz13] with permission from publisher of article under an open-access Creative Commons license).

2(e)(f)(g) show configurations of disclinations. Imagine cutting the cylinder with a half plane, rotating the cut surfaces by a vector $\boldsymbol{\omega}$, welding the cut surfaces together and relaxing (i.e. letting the body attain force equilibrium). Then a rotation discontinuity occurs on the cut surface and the vector $\boldsymbol{\omega}$ is called the Frank vector. If the Frank vector is parallel to the cylinder's axis, the disclination is called a wedge disclination; if the Frank vector is normal to the cylinder's axis, the line defect is called a twist disclination. In the following, we will mostly focus on wedge disclinations.

A wedge disclination can be visualized easily [Naz13], as shown in Figure 3. By taking away or inserting a wedge of an angle ω , a positive or negative wedge disclination is formed. In Figure 3(a) is a negative wedge disclination and (b) is a positive wedge disclination in a cylindrical body. After eliminating the overlap/gap-wedge and welding and letting the body relax, the body is in a state of internal stress corresponding to that of the wedge disclination (of corresponding sign).

In this work, we introduce a description for the disclination configuration based on the elastic distortion field, as shown in Figure 4. In Figure 4, red lines represent one elastic distortion field (possibly represented by the Identity tensor); black lines represent another distortion field. Thus, there is a surface of discontinuity between these two distortion fields and a terminating line (which is a point on the 2-d plane) on the interface is called a disclination. Also, there is a gap-wedge between the red part and the black part as shown in Figure 4(a), indicating it as a positive disclination; an overlap-wedge in Figure 4(b) corresponds to a negative disclination. Since a gap-wedge is eliminated for a positive disclination, there is circumferential tension around the core. Similarly, there is circumferential compression around the core for the negative disclination because of the inserted wedge. These physical arguments allow the inference of some features of the internal state of stress around disclination defects without further calculation.

A disclination loop is formed if an inclusion of a crystal with one orientation, and in the shape of



(a) A positive wedge disclination with a gap-wedge between two orientations. The red dot is the positive wedge disclination core where the interface of the orientation-discontinuity terminates. (b) A negative wedge disclination with an overlapwedge between two orientations. The green dot is the negative wedge disclination core where the interface of the orientation-discontinuity terminates.

Figure 4: An elastic distortion based description of wedge disclinations.



Figure 5: A 3D description for a disclination loop in an infinite block. ABCD is the disclination loop in a parallelepiped. Wedge disclinations exist along AB and CD while twist disclinations exist along AD and BC.

a parallelepiped with infinite length, is inserted in another infinite crystal of a different orientation, as shown in Figure 5. Focusing on the bottom surface of the parallelepiped, we consider the 'exterior' crystal as having one set of atomic planes parallel to the y - z plane bounded by unbounded black rectangles in Figure 5. The interior crystal has one set of planes at an angle of α to the y - z plane. The line of intersection AB represents a termination of a gap-wedge formed by the red plane of the interior crystal and the plane ABFE of the exterior crystal. Because the misorientation vector is directed along line AB (z axis), the latter serves as a wedge disclination. Similarly, there is a wedge disclination along intersection line CD of opposite sign to AB. For intersection lines BCand DA, the misorientation vector is perpendicular to the direction of intersection lines and twist disclinations of opposite signs are formed along BC and DA. The curve ABCD forms a disclination loop in the body (on elimination of the gap and overlap wedges).

4.2 Disclination dipole formation and movement in a lattice

Due to the addition and subtraction of matter over large distances involved in the definition of a disclination in the interior of a body, it is intuitively clear that a single disclination should cause long-range elastic stresses, which can also be seen from the analytical solution given in Section 2. Thus, a disclination rarely exists alone. Instead, usually, disclinations appear in pairs in the form of dipoles, namely a pair(s) of disclinations with opposite signs. Figure 6 shows a schematic of how a disclination dipole can form in a hexagonally coordinated structure. Figure 6(a) is the original structure with a hexagonal lattice; Figure 6(b) shows how bonds can be broken and rebuilt to transform a hexagon pair to a pentagon-heptagon pair in a topological sense (this may be thought of as a situation before relaxation); Figure 6(c) presents the relaxed configuration with a disclination dipole (the penta-hepta pair) after the transformation.

In a stress-free hexagonal lattice, removing an edge of a regular hexagon to form a pentagon can be associated with forming a positive wedge disclination at the center of the regular polygon (due to the tensile stress created in the circumferential direction); similarly, adding an edge to form a heptagon may be considered the equivalent of forming a negative wedge disclination. Hence, a heptagon-pentagon pair in a nominally hexagonal lattice is associated with a disclination dipole. It should be clear by the same logic that in a lattice with regular *n*-sided repeat units, an (n-1)-(n+1)polygon pair may be viewed as a disclination dipole.

Figure 7 shows how a disclination dipole moves by local crystal rearrangement under some external force. Figure 7(a) shows the configuration for a hexagonal lattice with a disclination dipole; then some atomic bonds nearby are broken and rebuilt in Figure 7(b); Figure 7(c) shows the relaxed configuration, where the disclination dipole has moved the right. The movement of a disclination dipole is a local rearrangement instead of a global rearrangement required to move a single disclination.

4.3 Descriptions of a dislocation by a (g.)disclination dipole

In this section we consider two physically distinct constructions that motivate why a straight edge dislocation may be thought of as being closely related to a (g.)disclination dipole. Figure 8(a) is a perfect crystal structure. Black lines represent atomic planes and red lines are the atomic bonds between two horizontal atomic planes. We apply a shear on the top and the bottom of this body along the blue arrows shown in Figure 8(a). After shearing, an extra half plane is introduced in the bottom part, as shown in Figure 8(b). This dislocation can as well be interpreted as a disclination dipole; a positive disclination (the red dot in Figure 8(c)) exists in the top part and a negative disclination (the green dot in Figure 8(c)) exists in the bottom part. Figure 8(d) shows a zoomed-



(a) Structure of a hexagonal lattice.



(b) Break and rebuild atomic bonds to form a disclination dipole (pentagon-heptagon pair).



(c) Relaxed configuration with a disclination dipole.

Figure 6: Kinematics of formation of a disclination-dipole in a hexagonal lattice. Figures constructed with *Chemdoodle*[che].



(a) Structure of a hexagonal lattice with a disclination dipole.



(b) Break and rebuild atomic bonds to move a disclination dipole.



(c) Relaxed configuration with disclination dipole having moved through the material to the right.

Figure 7: Kinematics of motion of a disclination dipole. Figures constructed with *Chemdoodle*[che].



(a) A perfect crystal structure, where the black lines represent atomic planes.



(b) Half-planes of atoms (1 - 7) in the top-block change topological connections to their counterparts in the bottom block on shearing, resulting in the appearance of an 'extra' half-plane in the bottom block. No extra atoms are introduced in the structure.



(c) Interpretation of defected structure as a disclination dipole. The red and green dots represent positive and negative wedge disclinations, respectively.



(d) Disclination dipole in (c) viewed at a larger length scale (weaker resolution). The disclination dipole appears as an edge dislocation.

Figure 8: Interpretation of a wedge disclination dipole as an edge dislocation.

out macroscopic view of the final configuration with an extra half-plane (of course obtained by a process where no new atoms have been introduced). Thus, a dislocation can be represented as a disclination dipole with very small separation distance. The Burgers vector of the dislocation is determined by the misorientation of the disclinations as well as the interval distance, as discussed in detail in Section 8. The upper disclination has a gap-wedge, namely a positive disclination, while the lower disclination has an overlap-wedge which is a negative disclination. Thus, the upper part is under tension and the bottom part under compression, consistent with the dislocation description with an 'extra' half-plane in the bottom part. Our rendition here is a way of understanding how a two-line, two-rotation axes disclination dipole [RK09] results in an edge dislocation in the limit of the distance between the two planes vanishing.

Another way in which a dislocation can be associated with a disclination dipole is one that is related to the description of incoherent grain boundaries. Figure 9(a) is an incompatible grain boundary represented by orientation fields, where black and red lines represent two different orientations. In Figure 9(b), the grain boundary interface is cut in two parts and the cut points are treated as a disclination dipole; the red dot is the positive disclination and the green dot is the negative disclination. In contrast to the description in Fig. 8, here the discontinuity surfaces being terminated by the disclinations are coplanar. In Figure 9(b), the disclination on the left is of negative strength while the disclination on the right is positive. It is to be physically expected that the disclination on the left of the dipole produces a compressive stress field in the region to the left of the dipole. Similarly, the disclination on the right of the pair should produce a tensile stress field to the right of the dipole. Figure 9(c) is the stress field for the grain boundary in Figure 9(a) modeled by a single disclination dipole through a numerical approximation of a theory to be described in Section 6. Indeed, the calculation bears out the physical expectation - the blue part represents a region with compressive stress and the red part a region with tensile stress. The stress field may be associated with that of an edge dislocation with Burgers vector in the vertical direction with an extra half plane of atoms in the right-half plane of the figure. This description of a dislocation by a disclination dipole is a way of understanding a single-line, two-rotation-axes dipole [RK09].

4.4 Grain boundaries via (g.)disclinations

We have already seen in the last section that a disclination-dipole model can be relevant to modeling the geometry and mechanics of grain boundaries. Figure 10 motivates how disclination dipoles arise naturally in the idealized description of a grain boundary from a microscopic view. In Figure 10(a), there are two grains with different orientations. After putting these two grains together and connecting the adjacent atomic bonds, we form a grain boundary, as shown in Figure 10(b). There exists a series of disclination dipoles along the boundary, as shown in Figure 10(b) where a blue pentagon is a negative disclination while a red triangle is a positive disclination.

In some cases, grain boundaries involve other types of defects beyond disclination dipoles, such as dislocations. Figure 11 is an example of a vicinal crystal interface [BAC05], which consists of a combination of dislocations and disclinations along the interface. In Figure 11(a) a high-angle tilt boundary with a tilt of 53.1° is viewed along the < 100 > tilt axis. If we slightly increase the tilt angle while keeping the topology of bond connections near the boundary fixed, high elastic deformations are generated, as shown in the Figure 11(b). Instead, an array of dislocations is often observed along the boundary as shown in the configuration Figure 11(c), presumably to eliminate long-range elastic deformations. In [ZAP16], we calculate the elastic fields of such boundaries utilizing both g.disclinations and dislocations.



(a) A defected grain boundary.



(b) A disclination dipole representing one defect of the grain boundary.



(c) Stress σ_{yy} around a single defect in the grain boundary, calculated from the (g.)disclination dipole model.





(a) Two grains with different crystal structures.

(b) Bi-crystal after merging two grains together.

Figure 10: Schematic of forming a series of disclination dipoles in a grain boundary. The red triangle is a positive disclination while the blue pentagon is a negative disclination. The pentagon-triangle disclination dipoles (in a 4-coordinated medium) exist along the interface.



Figure 11: (a) A common tilt grain boundary with 53.1 degree misorientation. (b) The configuration after applying additional tilt angle without any rearrangement. (c) The configuration with some dislocations introduced along the interface to eliminate far field deformation. (Figures reproduced from [BAC05] with permission from John Wiley and Sons).

4.4.1 Relationship between the disclination and dislocation models of a low-angle grain boundary

Normally, a grain boundary is modeled by an array of dislocations. As discussed in Section 2, a dislocation model cannot deal with a high-angle grain boundary. An alternative is to interpret the grain boundary through a disclination model as we discussed above. In this section, the relation between the disclination and dislocation models for a low-angle grain boundary is explained.

Consider a defect-free crystal as shown in Figure 12(a). First, we horizontally cut the material into four parts, as shown in the upper configuration in Figure 12(b). Now, for every part, we cut the material along its center surface (the dashed line in Figure 12(b)), insert one atom at the top and take away one atom from the bottom, weld the two half parts together again and relax the material. Then the configuration for every part will become the configuration at the bottom in Figure 12(b). By inserting and taking away atoms, we generate a negative disclination at the top and a positive disclination at the bottom. Repeating the same procedure for the remaining three parts, we finally obtain four parts with configurations as in Figure 12(c). The blue pentagons are negative disclinations while the red triangles are positive disclinations; a pair of a blue pentagon and an adjacent red triangle forms a disclination dipole. Next, we weld back these four parts and relax the whole material. Finally, a crystal configuration as in Figure 12(d) is generated, which is a grain boundary with the boundary interface shown as the blue dashed line. Along the grain boundary, dislocations exist along the interface with extra atomic planes shown as red lines in Figure 12(d). When the pentagon-triangle (5-3) disclination dipole is brought together to form a dislocation, the pentagon-triangle structure actually disintegrates and becomes a pentagon-square-square (5-4-4) object. Thus, a grain boundary can be constructed from a series of disclination dipoles; at the same time, we can see dislocation structures at the grain boundary interface. It is as if the 5-3 disclination dipole structure fades into the dislocation structure on coalescing the two disclinations in a dipole.

A comparison between the dislocation model and the disclination model has also been discussed in [Li72] where the possibility of modeling a dislocation by a disclination dipole is proposed within





(a) The perfect crystal configuration.

(b) Cut the material into four parts; introduce a positive disclination at the top and a negative disclination at the bottom.







(d) Weld four parts together and form a grain boundary whose interface is shown as the dotted blue line.



the context of the theory of linear elasticity. In this paper, we have elucidated the physical picture of forming a dislocation from a disclination dipole and, in subsequent sections, we also derive the general relationship between the Burgers vector of a dislocation and the disclination dipole for both the small and finite deformation cases, capitalizing crucially on a g.disclination formulation of a disclination dipole.

5 The Burgers vector of a disclination dipole in linear elasticity

We derive a formula for the Burgers vector of a wedge disclination dipole utilizing the linear theory of plane isotropic elasticity. Consider a positive disclination located at the origin O, as shown in Figure 13(a). We denote the stress at point c of a single disclination located at a with Frank vector Ω as $\sigma(c; a, \Omega)$. Thus, the stress field at r in Figure 13(a) is $\sigma(r; 0, \Omega)$. Next we consider the field point $r + \delta r$ marked by the green point as in Figure 13(b), with the disclination kept at the origin O. The stress tensor at this point is given by $\sigma(r + \delta r; 0, \Omega)$.

Instead of moving the field point in Figure 13(b), we next consider the field point as fixed at \boldsymbol{r} with the disclination moved from $\boldsymbol{0}$ to $-\delta \boldsymbol{r}$ as shown in Figure 13(c). The value of the stress at \boldsymbol{r} now is $\boldsymbol{\sigma}(\boldsymbol{r}; -\delta \boldsymbol{r}, \boldsymbol{\Omega})$. Utilizing the results in [DeW73a], the stress of a disclination, of fixed strength $\boldsymbol{\Omega}$ and located at $\boldsymbol{a} = a_1\boldsymbol{e}_1 + a_2\boldsymbol{e}_2 + a_3\boldsymbol{e}_3$, at the field point $\boldsymbol{c} = c_1\boldsymbol{e}_1 + c_2\boldsymbol{e}_2 + c_3\boldsymbol{e}_3$ is given by

$$\begin{split} \sigma_{11}(\boldsymbol{c};\boldsymbol{a},\boldsymbol{\Omega}) &= -\frac{G\Omega_{1}(c_{3}-a_{3})}{2\pi(1-\nu)} \left[\frac{c_{1}-a_{1}}{\rho^{2}} - 2\frac{(c_{1}-a_{1})(c_{2}-a_{2})^{2}}{\rho^{4}} \right] \\ &\quad -\frac{G\Omega_{2}(c_{3}-a_{3})}{2\pi(1-\nu)} \left[\ln\rho + \frac{(c_{2}-a_{2})^{2}}{\rho^{2}} + \frac{\nu}{1-2\nu} \right] \\ \sigma_{22}(\boldsymbol{c};\boldsymbol{a},\boldsymbol{\Omega}) &= -\frac{G\Omega_{2}(c_{3}-a_{3})}{2\pi(1-\nu)} \left[\frac{c_{1}-a_{1}}{\rho^{2}} + 2\frac{(c_{1}-a_{1})(c_{2}-a_{2})^{2}}{\rho^{4}} \right] \\ &\quad -\frac{G\Omega_{2}(c_{3}-a_{3})}{2\pi(1-\nu)} \left[\frac{c_{2}-a_{2}}{\rho^{2}} - 2\frac{(c_{1}-a_{1})^{2}(c_{2}-a_{2})}{\rho^{4}} \right] \\ &\quad + \frac{G\Omega_{3}}{2\pi(1-\nu)} \left[\ln\rho + \frac{(c_{1}-a_{1})^{2}}{\rho^{2}} + \frac{\nu}{1-2\nu} \right] \end{split}$$
(1)
 \\ \sigma_{33}(\boldsymbol{c};\boldsymbol{a},\boldsymbol{\Omega}) &= -\frac{G\nu(c_{3}-a_{3})}{\pi(1-\nu)\rho^{2}} (\Omega_{1}(c_{1}-a_{1}) + \Omega_{2}(c_{2}-a_{2})) + \frac{G\Omega_{3}}{2\pi(1-\nu)} \left[2\nu\ln\rho + \frac{\nu}{1-2\nu} \right] \\ \sigma_{12}(\boldsymbol{c};\boldsymbol{a},\boldsymbol{\Omega}) &= \frac{G\Omega_{1}(c_{3}-a_{3})}{2\pi(1-\nu)} \left[\frac{c_{2}-a_{2}}{\rho^{2}} - 2\frac{(c_{1}-a_{1})^{2}(c_{2}-a_{2})}{\rho^{4}} \right] \\ &\quad + \frac{G\Omega_{2}(c_{3}-a_{3})}{\pi(1-\nu)\rho^{2}} \left[\frac{c_{1}-a_{1}}{\rho^{2}} - 2\frac{(c_{1}-a_{1})^{2}(c_{2}-a_{2})}{\rho^{4}} \right] \\ \sigma_{23}(\boldsymbol{c};\boldsymbol{a},\boldsymbol{\Omega}) &= \frac{G\Omega_{1}(c_{3}-a_{3})}{2\pi(1-\nu)} \left[\frac{c_{1}-a_{1}}{\rho^{2}} - 2\frac{(c_{1}-a_{1})^{2}(c_{2}-a_{2})}{\rho^{4}} \right] \\ \sigma_{23}(\boldsymbol{c};\boldsymbol{a},\boldsymbol{\Omega}) &= \frac{G\Omega_{1}(c_{1}-a_{1})(c_{2}-a_{2})}{2\pi(1-\nu)\rho^{2}} - \frac{G\Omega_{2}}{2\pi(1-\nu)} \left[(1-2\nu)\ln\rho + \frac{(c_{1}-a_{1})^{2}}{\rho^{2}} \right] \\ \sigma_{13}(\boldsymbol{c};\boldsymbol{a},\boldsymbol{\Omega}) &= -\frac{G\Omega_{1}}{2\pi(1-\nu)} \left[(1-2\nu)\ln\rho + \frac{(c_{2}-a_{2})^{2}}{\rho^{2}} \right] + \frac{G\Omega_{2}(c_{1}-a_{1})(c_{2}-a_{2})}{2\pi(1-\nu)\rho^{2}}, \end{split}

where ρ is the distance between the field point c and the source point a, $\rho = |c - a|$. Equation (1) shows that the stress fields only depend on the relative location of the field and disclination source

points. In other words, given a disclination at a with Frank vector Ω , the stress field at point c can be expressed as

$$\boldsymbol{\sigma}(\boldsymbol{c};\boldsymbol{a},\boldsymbol{\Omega})=\boldsymbol{f}(\boldsymbol{c}-\boldsymbol{a};\boldsymbol{\Omega}),$$

where f is the formula for the stress field of the wedge disclination in linear isotropic elasticity whose explicit expression in Cartesian coordinates is given in (1). From (1), we have

$$\boldsymbol{f}(\boldsymbol{x};\boldsymbol{\Omega}) = -\boldsymbol{f}(\boldsymbol{x};-\boldsymbol{\Omega}),\tag{2}$$

for any given \boldsymbol{x} . Thus, for the stress fields, we have

$$\boldsymbol{\sigma}(\boldsymbol{c};\boldsymbol{a},\boldsymbol{\Omega}) = -\boldsymbol{\sigma}(\boldsymbol{c};\boldsymbol{a},-\boldsymbol{\Omega}).$$

Hence, the stress field in Figure 13(a) can be written as

$$\boldsymbol{\sigma}(\boldsymbol{r};\boldsymbol{0},\boldsymbol{\Omega}) = \boldsymbol{f}(\boldsymbol{r};\boldsymbol{\Omega}).$$

The stress field corresponding to Figure 13(b) is

$$\boldsymbol{\sigma}(\boldsymbol{r}+\delta \boldsymbol{r};\boldsymbol{0},\boldsymbol{\Omega})=\boldsymbol{f}(\boldsymbol{r}+\delta \boldsymbol{r};\boldsymbol{\Omega})$$

Also, the stress field in Figure 13(c) is

$$\boldsymbol{\sigma}(\boldsymbol{r};-\delta\boldsymbol{r},\boldsymbol{\Omega}) = \boldsymbol{f}(\boldsymbol{r}-(-\delta\boldsymbol{r});\boldsymbol{\Omega}) = \boldsymbol{f}(\boldsymbol{r}+\delta\boldsymbol{r};\boldsymbol{\Omega}).$$

In Figure 13(d), a disclination dipole is introduced. A negative disclination with Frank vector $-\Omega$ is at **0** and the positive disclination with Frank vector Ω is at $-\delta \mathbf{r}$. Thus, $\delta \mathbf{r}$ is the separation vector of the dipole, pointing from the positive disclination to the negative disclination and we are interested in calculating the stress at \mathbf{r} , represented as the red dot. Let the stress field for the disclination configuration in Figure 13(d) be denoted as $\hat{\boldsymbol{\sigma}}$. Due to superposition in linear elasticity, the stress field of Figure 13(d) can be written as

$$egin{aligned} \hat{\pmb{\sigma}}(\pmb{r};\delta\pmb{r},\pmb{\Omega}) &:= \pmb{\sigma}(\pmb{r};-\delta\pmb{r},\pmb{\Omega}) + \pmb{\sigma}(\pmb{r};\pmb{0},-\pmb{\Omega}) \ \Rightarrow \hat{\pmb{\sigma}}(\pmb{r};\delta\pmb{r},\pmb{\Omega}) &= \pmb{f}(\pmb{r}+\delta\pmb{r};\pmb{\Omega}) + \pmb{f}(\pmb{r};-\pmb{\Omega}). \end{aligned}$$

On applying (2), we have

$$\hat{\boldsymbol{\sigma}}(\boldsymbol{r};\delta\boldsymbol{r},\boldsymbol{\Omega}) = \boldsymbol{f}(\boldsymbol{r}+\delta\boldsymbol{r};\boldsymbol{\Omega}) - \boldsymbol{f}(\boldsymbol{r};\boldsymbol{\Omega})$$
$$\Rightarrow \hat{\boldsymbol{\sigma}}(\boldsymbol{r};\delta\boldsymbol{r},\boldsymbol{\Omega}) = \boldsymbol{\sigma}(\boldsymbol{r}+\delta\boldsymbol{r};\boldsymbol{0},\boldsymbol{\Omega}) - \boldsymbol{\sigma}(\boldsymbol{r};\boldsymbol{0},\boldsymbol{\Omega}).$$
(3)

Therefore, we have shown that the stress field in Figure 13(d) equals the difference between the stress fields in Figure 13(b) and the one in Figure 13(a).

Specializing to the plane case with $\mathbf{r} = x_1 \mathbf{e}_1 + x_2 \mathbf{e}_2$, the stress field corresponding to Figure 13(a), given the Frank vector $\boldsymbol{\Omega} = \Omega_3 \mathbf{e}_3$, is

$$\sigma_{11}(\mathbf{r};\mathbf{0},\boldsymbol{\Omega}) = f_{11}(x_1, x_2;\Omega_3) = \frac{G\Omega_3}{2\pi(1-\nu)} \left[\ln r + \frac{x_2^2}{r^2} + \frac{\nu}{1-2\nu} \right]$$

$$\sigma_{22}(\mathbf{r};\mathbf{0},\boldsymbol{\Omega}) = f_{22}(x_1, x_2;\Omega_3) = \frac{G\Omega_3}{2\pi(1-\nu)} \left[\ln r + \frac{x_1^2}{r^2} + \frac{\nu}{1-2\nu} \right]$$

$$\sigma_{12}(\mathbf{r};\mathbf{0},\boldsymbol{\Omega}) = f_{12}(x_1, x_2;\Omega_3) = -\frac{G\Omega_3 x_1 x_2}{2\pi(1-\nu)r^2},$$



(a) A positive disclination located at the coordinate origin with a field point located at \boldsymbol{r} .



(c) The configuration with disclination source moved to $-\delta \boldsymbol{r}$.

(d) Place a disclination dipole with the separation vector δr and keep the field point at r.

Figure 13: Schematic in support of calculation of stress field of a wedge-disclination dipole in linear, plane, isotropic elasticity.

where r is the norm of r, G is the shear modulus and ν is the Poisson ratio. Assuming

$$\delta \boldsymbol{r} := \delta x_1 \boldsymbol{e}_1 + \delta x_2 \boldsymbol{e}_2 \tag{4}$$

to be small, the Taylor expansion of $\boldsymbol{\sigma}(\boldsymbol{r} + \delta \boldsymbol{r}; \boldsymbol{0}, \boldsymbol{\Omega})$ is

$$\boldsymbol{\sigma}(\boldsymbol{r}+\delta \boldsymbol{r};\boldsymbol{0},\boldsymbol{\Omega})=\boldsymbol{f}(\boldsymbol{r};\boldsymbol{\Omega})+rac{\partial \boldsymbol{f}(\boldsymbol{r};\boldsymbol{\Omega})}{\partial \boldsymbol{r}}\delta \boldsymbol{r}+O(\delta \boldsymbol{r}^{2}).$$

After substituting $\mathbf{r} = x_1 \mathbf{e}_1 + x_2 \mathbf{e}_2$ and $\delta \mathbf{r} = \delta x_1 \mathbf{e}_1 + \delta x_2 \mathbf{e}_2$, we have

$$\sigma_{11}(\mathbf{r} + \delta \mathbf{r}; \mathbf{0}, \mathbf{\Omega}) = f_{11}(x_1, x_2; \Omega_3) + \frac{\partial f_{11}(x_1, x_2; \Omega_3)}{\partial x_1} \delta x_1 + \frac{\partial f_{11}(x_1, x_2; \Omega_3)}{\partial x_2} \delta x_2 + O(\delta \mathbf{r}^2)$$

$$\sigma_{22}(\mathbf{r} + \delta \mathbf{r}; \mathbf{0}, \mathbf{\Omega}) = f_{22}(x_1, x_2; \Omega_3) + \frac{\partial f_{22}(x_1, x_2; \Omega_3)}{\partial x_1} \delta x_1 + \frac{\partial f_{22}(x_1, x_2; \Omega_3)}{\partial x_2} \delta x_2 + O(\delta \mathbf{r}^2)$$

$$\sigma_{12}(\mathbf{r} + \delta \mathbf{r}; \mathbf{0}, \mathbf{\Omega}) = f_{12}(x_1, x_2; \Omega_3) + \frac{\partial f_{12}(x_1, x_2; \Omega_3)}{\partial x_1} \delta x_1 + \frac{\partial f_{12}(x_1, x_2; \Omega_3)}{\partial x_2} \delta x_2 + O(\delta \mathbf{r}^2).;$$

After substituting $\sigma(r; \mathbf{0}, \Omega)$ and $\sigma(r + \delta r; \mathbf{0}, \Omega)$ into (3) and omitting the higher order terms, we get

$$\hat{\sigma}_{11}(x_1, x_2; \delta x_1, \delta x_2, \Omega_3) = \frac{G\Omega_3 \delta x_2}{2\pi (1 - \nu)} \left[\frac{x_2}{r^2} + 2\frac{x_1^2 x_2}{r^4} \right] + \frac{G\Omega_3 \delta x_1}{2\pi (1 - \nu)} \left[\frac{x_1}{r^2} - 2\frac{x_1 x_2^2}{r^4} \right]$$

$$\hat{\sigma}_{22}(x_1, x_2; \delta x_1, \delta x_2, \Omega_3) = \frac{G\Omega_3 \delta x_2}{2\pi (1 - \nu)} \left[\frac{x_2}{r^2} - 2\frac{x_1^2 x_2}{r^4} \right] + \frac{G\Omega_3 \delta x_1}{2\pi (1 - \nu)} \left[\frac{x_1}{r^2} + 2\frac{x_1 x_2^2}{r^4} \right]$$

$$\hat{\sigma}_{12}(x_1, x_2; \delta x_1, \delta x_2, \Omega_3) = -\frac{G\Omega_3 \delta x_2}{2\pi (1 - \nu)} \left[\frac{x_2}{r^2} - 2\frac{x_1 x_2^2}{r^4} \right] - \frac{G\Omega_3 \delta x_1}{2\pi (1 - \nu)} \left[\frac{x_2}{r^2} - 2\frac{x_1^2 x_2}{r^4} \right].$$
(5)

The stress field of the single edge dislocation in 2-D, isotropic elasticity is [DeW73b]

$$\sigma_{11}^{b}(x_{1}, x_{2}; b_{1}, b_{2}) = -\frac{Gb_{1}}{2\pi(1-\nu)} \left[\frac{x_{2}}{r^{2}} + 2\frac{x_{1}^{2}x_{2}}{r^{4}} \right] + \frac{Gb_{2}}{2\pi(1-\nu)} \left[\frac{x_{1}}{r^{2}} - 2\frac{x_{1}x_{2}^{2}}{r^{4}} \right]$$

$$\sigma_{22}^{b}(x_{1}, x_{2}; b_{1}, b_{2}) = -\frac{Gb_{1}}{2\pi(1-\nu)} \left[\frac{x_{2}}{r^{2}} - 2\frac{x_{1}^{2}x_{2}}{r^{4}} \right] + \frac{Gb_{2}}{2\pi(1-\nu)} \left[\frac{x_{1}}{r^{2}} + 2\frac{x_{1}x_{2}^{2}}{r^{4}} \right]$$

$$\sigma_{12}^{b}(x_{1}, x_{2}; b_{1}, b_{2}) = \frac{Gb_{1}}{2\pi(1-\nu)} \left[\frac{x_{2}}{r^{2}} - 2\frac{x_{1}x_{2}^{2}}{r^{4}} \right] - \frac{Gb_{2}}{2\pi(1-\nu)} \left[\frac{x_{2}}{r^{2}} - 2\frac{x_{1}^{2}x_{2}}{r^{4}} \right].$$
(6)

On defining the Burgers vector of a disclination dipole with separation vector δr (4) and strength Ω as

$$\boldsymbol{b} := -\Omega_3 \delta r_2 \boldsymbol{e}_1 + \Omega_3 \delta r_1 \boldsymbol{e}_2 = \boldsymbol{\Omega} \times \delta \boldsymbol{r},\tag{7}$$

we see that the stress field of the disclination dipole (5) exactly matches that of the single edge dislocation (6).

This establishes the correspondence between the Burgers vector of the wedge disclination dipole and the edge dislocation in 2-d, isotropic, plane, linear elasticity. In Section 8 we establish the general form of this geometric relationship in the context of exact kinematics, valid for any type of material (i.e. without reference to material response).



Figure 14: Physical regularization of classical terminating discontinuity. Treat the distortion discontinuity as the eigenwall field S with support in a layer.

6 Generalized disclination theory and associated Weingarten's theorem

The connection between g.disclinations (and dislocations) represented as fields and their more classical representation following Weingarten's pioneering work is established in Sections 7 and 8. In this Section we briefly review the defect kinematics of g.disclination theory and a corresponding Weingarten-gd theorem developed in [AF15] that are necessary prerequisites for the arguments in the aforementioned sections. We also develop a new result in Section 6.2 related to the Weingarten-gd theorem, proving that the inverse deformation jump across the cut-surface is independent of the surface when the g.disclination density vanishes.

As defined in Section 2, a single g.disclination is a line defect terminating a distortion discontinuity. Developed as a generalization of eigendeformation theory of Kroner, Mura and deWit, the generalized disclination has a core and the discontinuity is modeled by an eigenwall field with support in a layer [AF15], as shown in Figure 14. The representation of a discrete g.disclination involves a continuous elastic 2-distortion field \mathbf{Y} , assumed to be irrotational outside the generalized disclination core ($\mathbf{Y} = \text{grad grad}(\mathbf{x}^{-1})$ in the case without defects, where \mathbf{x} is the deformation map). The strength of the discrete generalized disclination is given by the second order tensor obtained by integrating the 2-distortion field along any closed curve encircling the core; when defined from a terminating distortion discontinuity, it is simply the difference of the two distortions involved in defining the discontinuity. One way of setting up the generalized disclination density tensor field, which is a third order tensor, is to assign the tensor product of the strength tensor and the core line direction vector as a uniformly distributed field within the generalized disclination core, and zero outside it. In the case of a disclination, the strength tensor is necessarily the difference of two orthogonal tensors.

The fundamental kinematic decomposition of generalized disclination theory [AF15] is to write

$$Y = \text{grad } W + S, \tag{8}$$

where W is the i-elastic 1-distortion (F^{-1} in the defect-free case, where F is the deformation gradient) and S (3^{rd} -order tensor) is the eigenvall field.

With this decomposition of Y, it is natural to measure the generalized disclination density as

$$\operatorname{curl}\left(\boldsymbol{Y} - \operatorname{grad}\boldsymbol{W}\right) = \operatorname{curl}\boldsymbol{S} =: \boldsymbol{\Pi}.$$
(9)

It characterizes the closure failure of integrating \boldsymbol{Y} on closed contours in the body:

$$\int_{a} \boldsymbol{\Pi} \boldsymbol{n} da = \int_{c} \boldsymbol{Y} d\boldsymbol{x}$$
(10)

where a is any area patch with closed boundary contour c in the body. Physically, it is to be interpreted as a density of lines (threading areas) in the current configuration, carrying a tensorial attribute that reflects a jump in W.

The dislocation density is defined as [AF15]

$$\boldsymbol{\alpha} := \boldsymbol{Y} : \boldsymbol{X} = (\boldsymbol{S} + \operatorname{grad} \boldsymbol{W}) : \boldsymbol{X}.$$
(11)

In the case that there is no distortion discontinuity, namely S = 0, (11) becomes $\alpha = -\operatorname{curl} W$, since $\operatorname{curl} A = -\operatorname{grad} A : X$ for any smooth tensor field A. The definition of the dislocation density (11) is motivated by the displacement-jump formula (16) [AF15] corresponding to a single, isolated defect line terminating an i-elastic distortion jump in the body. In this situation, the displacement jump for an isolated defect line, measured by integrating W along any closed curve encircling the defect core cylinder³, is no longer a topological object independent of the curve (in the class of curves encircling the core) due to the fact that in the presence of a g.disclination density localized in the core cylinder the field S cannot be localized in the core - it is, at the very least, supported in a layer extending to the boundary from the core, or, when div S = 0, completely delocalized over the entire domain.

Now we apply a Stokes-Helmholtz-like orthogonal decomposition of the field S into a compatible part and an incompatible part:

$$S = S^{\perp} + \text{grad } Z^{s}$$

$$\operatorname{curl} S^{\perp} = \Pi$$

$$\operatorname{div} S^{\perp} = 0$$

with $S^{\perp} n = 0$ on the boundary.
(12)

It is clear that when $\Pi = 0$ then $S^{\perp} = 0$.

In summary, the governing equations for computing the elastic fields for static generalized disclination theory (i.e. when the disclination and dislocation fields are specified) are

$$\operatorname{curl} \boldsymbol{S} = \boldsymbol{\Pi}$$

$$\boldsymbol{S} = \boldsymbol{S}^{\perp} + \operatorname{grad} \boldsymbol{Z}^{s}$$

$$\operatorname{div} \boldsymbol{S}^{\perp} = \boldsymbol{0} \quad \text{with } \boldsymbol{S}^{\perp} \boldsymbol{n} = \boldsymbol{0} \text{ on the boundary}$$

$$\boldsymbol{\alpha} = (\boldsymbol{S} + \operatorname{grad} \boldsymbol{W}) : \boldsymbol{X},$$
(13)

where Π and α are specified from physical considerations. These equations are solved along with balance of linear and angular momentum involving Cauchy stresses and couple-stresses (with

³In [AF15] a typographical error suggests that the displacement jump is obtained by integrating α on area patches; α there should have been replaced by curl W.

constitutive assumptions) to obtain g.disclination and dislocation stress and couple stress fields. In the companion paper [ZAP16] we solve these equations along with

$$\operatorname{div} T = 0$$

with T representing the Cauchy stress as a function of W, and we ignore couple stresses for simplicity.

6.1 Review of Weingarten theorem associated with g.disclinations

In this section we provide an overview of the Weingarten-gd theorem for g.disclinations introduced in [AF15]. Figure 15 shows cross-sections of three dimensional multi-connected bodies with toroidal (Figure 15(a)) and through holes (Figure 15(d)). In both cases, the multi-connected body can be transformed into a simply-connected one by introducing a cut-surface. For the toroidal case, putting the cut-surface either from a curve on the external surface to a curve on the exterior surface of the torus (Figure 15(b)) or putting the cut-surface with bounding curve along the interior surface of the torus (Figure 15(c)) will make the multi-connected domain into a simply-connected domain. Similarly, the body with the through-hole can be cut by a surface extending from a curve on the external surface to the surface of the hole. Figures 15(b) and 15(e) result in topological spheres while Fig. 15(c) results in a topological sphere with a contained interior cavity. In terms of g.disclination theory, the holes are associated with the cores of the defect lines.

Given a continuously differentiable 3-order tensor field Y on the multi-connected domain such that \tilde{Y} is symmetric in the last two indices and curl $\tilde{Y} = 0$, the Weingarten-gd problem asks if there exists a vector field y on the cut-induced simply-connected domain such that

grad grad
$$\boldsymbol{y} = \boldsymbol{Y}$$
,

and a formula for the possible jump $[\![y]\!]$ of y across the cut-surface. Also, since \tilde{Y} is curl-free and continuously differentiable on the multi-connected domain, we can defined a field \tilde{W} such that

$$\operatorname{grad} \tilde{W} = \tilde{Y}$$

on the corresponding simply-connected domain.

In the following, we will assign a unit normal field to any cut-surface. For any point on the cut-surface, say A, we will denote by A^+ a point arbitrarily close to A from the region into which the normal at A points and as A^- a similar point from the region into which the negative normal points. For any smooth function, say f, defined on the (multi)-connected domain, we will define

$$\boldsymbol{f}^{+}(\boldsymbol{A}) := \lim_{\boldsymbol{A}^{+} \to \boldsymbol{A}} \boldsymbol{f}(\boldsymbol{A}^{+}) \text{ and } \boldsymbol{f}^{-}(\boldsymbol{A}) := \lim_{\boldsymbol{A}^{-} \to \boldsymbol{A}} \boldsymbol{f}(\boldsymbol{A}^{-}).$$
(14)

Consider a closed contour in Fig. 16 in the multi-connected domain starting and ending at A and passing through B as shown. In addition, also consider as the 'inner' and 'outer' closed contours the closed curves that remain by ignoring the overlapping segments, the inner closed contour passing through A and the outer through B. Then, because of the continuity of \tilde{Y} and its vanishing *curl*, the line integral of \tilde{Y} on the inner and outer closed contours must be equal and this statement holds for any closed contour enclosing the hole. The line integral of \tilde{Y} on the closed contour is defined as

$$\int_C \tilde{\boldsymbol{Y}} d\boldsymbol{x} =: \boldsymbol{\varDelta}$$



(a) The cross-section of a multi-connected body with a toroidal hole. The shaded gray area is the 'half-toroid'. The half-toroid is not shown in Figures 15(b) and 15(c).



(b) The multi-connected body becomes simply-connected after introducing the cut-surface.



(c) Another method to introduce the cut surface to make the multi-connected body simply-connected.



(d) The cross-section of a multiconnected body with a through hole.



(e) The cross-section of a simplyconnected body with a through hole and a cut surface.

Figure 15: The cross-sections of multi-connected bodies with a toroidal hole or a through hole, and their corresponding simply-connected bodies by introducing cut-surfaces.



Figure 16: A contour enclosing the core on the cross-section of the multi-connected domain. The contour passes through points A and B. (Figure reproduced from [AF15] with permission from Springer).

Now, considering the cut-surface passing through A, if we construct the corresponding \tilde{W} , say \tilde{W}_1 , then the jump of \tilde{W}_1 is given by

$$\llbracket \tilde{\boldsymbol{W}}_1
rbracket (\boldsymbol{A}) = \int\limits_{C(A^-,A^+)} \tilde{\boldsymbol{Y}} d\boldsymbol{x} = \boldsymbol{\Delta},$$

where $C(A^-, A^+)$ is the curve from the inner closed contour with the point A taken out and with start-point A^- and the end-point A^+ , as shown in Figure 16. Similarly, a different cut-surface passing through another point B can be introduced and the corresponding \tilde{W}_2 can be constructed with $[\![\tilde{W}_2]\!](B) = \Delta$. Since A, B and the cut surfaces are chosen arbitrarily, the jump of any of the functions $[\![\tilde{W}]\!]$ across their corresponding cut-surface takes the same value, independent of the invoked cut-surface and the point on the surface.

In addition, due to the symmetry in the last two indices of \tilde{Y} , curl \tilde{W} vanishes. Thus, a vector field y can be defined, on the relevant cut-induced simply-connected domain associated with the construction of \tilde{W} , such that

$$\operatorname{grad} \boldsymbol{y} = \boldsymbol{W}.\tag{15}$$

Now choose a point x_0 arbitrarily on the cut-surface. Let x be any other point on this cut-surface, as shown in Figure 17.

Since

$$egin{aligned} m{W}^- &= (\operatorname{grad} m{y})^- \ m{ ilde W}^+ &= (\operatorname{grad} m{y})^+ \,, \end{aligned}$$



Figure 17: Cross-section of a simply-connected domain induced by a cut-surface. The red path is from x_0^- to x^- and the blue path is from x_0^+ to x^+ .

then y at x across the cut-surface is

$$egin{aligned} m{y}^+(m{x}) &= m{y}^+(m{x}_0) + \int_{m{x}_0^+}^{m{x}^+} ilde{m{W}}^+(m{x}') dm{x}' \ m{y}^-(m{x}) &= m{y}^-(m{x}_0) + \int_{m{x}_0^-}^{m{x}^-} ilde{m{W}}^-(m{x}') dm{x}' \end{aligned}$$

(by working on paths from $x_0^{+/-}$ to $x^{+/-}$ and then taking limits as the paths approach the cutsurface). Then the jump of y, $[\![y]\!]$ can be derived as

$$\llbracket \boldsymbol{y} \rrbracket(\boldsymbol{x}) = \boldsymbol{y}^+(\boldsymbol{x}) - \boldsymbol{y}^-(\boldsymbol{x}) = \llbracket \boldsymbol{y} \rrbracket(\boldsymbol{x}_0) + \int_{\boldsymbol{x}_0}^{\boldsymbol{x}} \llbracket \tilde{\boldsymbol{W}} \rrbracket(\boldsymbol{x}') d\boldsymbol{x}'$$

Recall that $[\![\tilde{W}]\!](x') = \Delta$, which is independent of the cut-surface and the point x' on it. Therefore,

$$\llbracket \boldsymbol{y} \rrbracket(\boldsymbol{x}) = \llbracket \boldsymbol{y} \rrbracket(\boldsymbol{x}_0) + \boldsymbol{\Delta}(\boldsymbol{x} - \boldsymbol{x}_0).$$
(16)

Furthermore, it can be shown that the jump at any point \boldsymbol{x} on the cut-surface is independent of the choice of the base point \boldsymbol{x}_0 (on the cut-surface).

In addition, consider the case where $[\![\tilde{W}]\!] = 0$ - i.e. the defect line is a pure dislocation. Then the Burgers vector is defined as

$$\boldsymbol{b}(\boldsymbol{x}) := \llbracket \boldsymbol{y} \rrbracket(\boldsymbol{x}),$$

and from Eqn 16, given an arbitrarily fixed cut-surface, we have

$$\llbracket \boldsymbol{y} \rrbracket(\boldsymbol{x}) = \llbracket \boldsymbol{y} \rrbracket(\boldsymbol{x}_0) \tag{17}$$



Figure 18: Arbitrary path (shown on the cross-section) for construction of a continuous field W^{τ} on the simply-connected domain induced by a cut-surface τ when $\Delta = 0$.

where \boldsymbol{x} and \boldsymbol{x}_0 are arbitrarily chosen points on the cut-surface. Thus, it has been shown that for $\boldsymbol{\Delta} = \boldsymbol{0}$, the displacement jump is independent of location on a given cut-surface.

In the next Section 6.2, we furthermore show that the Burgers vector is independent of the choice of the cut-surface as well when $\Delta = 0$ and α is localized in the core.

6.2 Cut-surface independence in the Weingarten-gd theorem for $\Delta = 0$

We now prove that the jump (17) across a cut-surface in the Weingarten-gd theorem is independent of the choice of the surface when $\Delta = 0$.

By hypothesis, there is a continuous field \tilde{Y} in the multi-connected body with curl $\tilde{Y} = \mathbf{0}$ and $\tilde{Y}_{ijk} = \tilde{Y}_{ikj}$. Also, after introducing an arbitrary cut-surface, as in Figure 15(e), we can construct fields \tilde{W} and \tilde{y} such that grad(grad \tilde{y}) = \tilde{Y} and grad $\tilde{W} = \tilde{Y}$. Based on the Weingarten-gd theorem, we have on this arbitrary chosen cut-surface

$$\llbracket \tilde{\boldsymbol{y}}(\boldsymbol{x}) \rrbracket = \llbracket \tilde{\boldsymbol{y}}(\boldsymbol{x}_0) \rrbracket + \boldsymbol{\Delta}(\boldsymbol{x} - \boldsymbol{x}_0),$$

where $\boldsymbol{\Delta} = \oint \tilde{\boldsymbol{Y}} d\boldsymbol{x}$. Since $\boldsymbol{\Delta} = \boldsymbol{0}$, it is clear that

$$\llbracket \tilde{\boldsymbol{y}}(\boldsymbol{x}) \rrbracket = \llbracket \tilde{\boldsymbol{y}}(\boldsymbol{x}_0) \rrbracket.$$

The goal now is to prove that

$$\llbracket \widetilde{oldsymbol{y}}(oldsymbol{x})
rbracket =: oldsymbol{b}$$

where the vector \boldsymbol{b} is independent of the choice of the cut-surface and, hence, independent of \boldsymbol{x} on the cut-surface as well using results of Section 6.1.

Since the definition of \tilde{W} depends on the cut-surface, given a simply-connected domain induced by a cut-surface τ , we can express any such \tilde{W} , say W^{τ} , on the simply-connected domain as

$$W_{ij}^{\tau}(\boldsymbol{x}, \boldsymbol{x}^{0}, p) := \int_{\boldsymbol{x}^{0}}^{\boldsymbol{x}} E_{ijk} \, dx_{k} + W_{ij}^{\tau}(\boldsymbol{x}^{0})$$
(18)

where p is a curve from \mathbf{x}^0 to \mathbf{x} as shown in Figure 18, and $\mathbf{E} := \tilde{\mathbf{Y}}$ with the field $\tilde{\mathbf{Y}}$ satisfying the constraint $\oint \tilde{\mathbf{Y}} d\mathbf{x} = \mathbf{\Delta} = \mathbf{0}$. Since the line integral of \mathbf{E} on any closed loop is zero, \mathbf{W}^{τ} as defined is independent of path on the original multi-connected domain and hence thinking of the constructed \mathbf{W}^{τ} as a continuous function on it makes sense.



Figure 19: A closed loop in the cross-section of a multi-connected domain to justify independence of the Burgers vector from the circuit used to evaluate it for $\Delta = 0$.

Now with the constructed W^{τ} , we can define the line integral

$$oldsymbol{b}^{ au,p} := \oint\limits_p oldsymbol{W}^ au doldsymbol{x}$$

on a closed loop p enclosing the core.

We now show first that $b^{\tau,p}$ is independent of the loop used to define it. Since grad $W^{\tau} = E$ from the definition (18) and E is symmetric in the last two indices by hypothesis,

$$E_{ijk} = W_{ij,k}^{\tau}$$

$$\Rightarrow E_{ijk} - E_{ikj} = W_{ij,k}^{\tau} - W_{ik,j}^{\tau} = 0$$

$$\Rightarrow e_{mkj}(W_{ij,k}^{\tau} - W_{ik,j}^{\tau}) = 0$$

$$\Rightarrow \operatorname{curl} \boldsymbol{W}^{\tau} = \boldsymbol{0}$$

Thus, on the multi-connected domain, given any arbitrary closed loop as in Figure 19,

$$\oint \boldsymbol{W}^{\tau} d\boldsymbol{x} = \boldsymbol{0}$$

$$\int_{\text{inner}} \boldsymbol{W}^{\tau} d\boldsymbol{x} - \int_{\text{outer}} \boldsymbol{W}^{\tau} d\boldsymbol{x} + \int_{B}^{A} \boldsymbol{W}^{\tau} d\boldsymbol{x} + \int_{A}^{B} \boldsymbol{W}^{\tau} d\boldsymbol{x} = \boldsymbol{0}$$

where $\int_{inner} \mathbf{W}^{\tau} d\mathbf{x}$ is the integral along the inner loop anti-clockwise and $\int_{outer} \mathbf{W}^{\tau} d\mathbf{x}$ is the integral along the outer loop anti-clockwise. Therefore, since $\int_{B}^{A} \mathbf{W}^{\tau} d\mathbf{x} + \int_{A}^{B} \mathbf{W}^{\tau} d\mathbf{x} = \mathbf{0}$,

$$\int_{\text{inner}} \boldsymbol{W}^{\tau} d\boldsymbol{x} = \int_{\text{outer}} \boldsymbol{W}^{\tau} d\boldsymbol{x}.$$



Figure 20: The integration path on the cross-section of a simply-connected domain to calculate the jump $[\![y]\!]$ at x.

Thus, $b^{\tau,p}$ is independent of the loop path p and we will denote it as b^{τ} .

Now for a simply-connected domain induced by the cut-surface τ , given a W^{τ} , there exists (many) y^{τ} satisfying grad $y^{\tau} = W^{\tau}$; any such y^{τ} may be expressed as

$$oldsymbol{y}^ au(oldsymbol{x};oldsymbol{x}_0) = \int_{oldsymbol{x}_0}^{oldsymbol{x}} oldsymbol{W}^ au doldsymbol{x} + oldsymbol{y}^ au(oldsymbol{x}_0).$$

Then, with reference to Fig. 20, we have

$$oldsymbol{y}^{ au}(oldsymbol{x}^+) = oldsymbol{y}^{ au}(oldsymbol{x}_0) + \int_{oldsymbol{x}_0}^{oldsymbol{x}^+} oldsymbol{W}^{ au} doldsymbol{x}$$

 $\Rightarrow oldsymbol{y}^{ au}(oldsymbol{x}^+) = oldsymbol{y}^{ au}(oldsymbol{x}_0) + \int_{oldsymbol{x}_0}^{oldsymbol{x}^-} oldsymbol{W}^{ au} doldsymbol{x} + \int_{oldsymbol{x}_0}^{oldsymbol{x}^+} oldsymbol{W}^{ au} doldsymbol{x}$
 $\Rightarrow oldsymbol{y}^{ au}(oldsymbol{x}^+) = oldsymbol{y}^{ au}(oldsymbol{x}_0) + \int_{oldsymbol{x}_0}^{oldsymbol{x}^-} oldsymbol{W}^{ au} doldsymbol{x} + \int_{oldsymbol{x}_0}^{oldsymbol{x}^+} oldsymbol{W}^{ au} doldsymbol{x}$

and

$$m{y}^{ au}(m{x}^{-}) = m{y}^{ au}(m{x}_0) + \int_{m{x}_0\p_2}^{m{x}^{-}} m{W}^{ au} dm{x}.$$

Thus,

$$\llbracket \boldsymbol{y}^{\tau} \rrbracket = \int_{\substack{\boldsymbol{x}^-\\p_3}}^{\boldsymbol{x}^+} \boldsymbol{W}^{\tau} d\boldsymbol{x} = \boldsymbol{b}^{\tau}.$$
(19)

We note next that if τ and τ' are two cut-surfaces, (18) and the continuity of $\mathbf{W}^{\tau}, \mathbf{W}^{\tau'}$ imply that $\mathbf{W}^{\tau} - \mathbf{W}^{\tau'}$ is a constant tensor on the original multi-connected domain, and therefore (19) implies that $\mathbf{b}^{\tau} = \mathbf{b}^{\tau'} =: \mathbf{b}$, a constant vector independent of the cut-surface.

Therefore, we have shown that when $\Delta = 0$, the Burgers vector is cut-surface independent.



Figure 21: The dislocation and the generalized disclination densities are localized in a defect core.

7 Interpretation of the Weingarten theorem in terms of g.disclination kinematics

The Weingarten-gd theorem for generalized disclinations (16) was reviewed in Section 6.1. We recall that the jump of the inverse-deformation \boldsymbol{y} across a cut-surface is characterized, in general, by the jump at an arbitrarily chosen point on the surface, $[\![\boldsymbol{y}(\boldsymbol{x}_0)]\!]$, and $\boldsymbol{\Delta} = [\![\tilde{\boldsymbol{W}}]\!]$, and all of these quantities are defined from the knowledge of the field $\tilde{\boldsymbol{Y}}$. However, given a g.disclination and a dislocation distribution $\boldsymbol{\Pi}$ and $\boldsymbol{\alpha}$, respectively, on the body, it is natural to ask as to what ingredients of g.disclination theory correspond to a candidate $\tilde{\boldsymbol{Y}}$ field. A consistency condition we impose is that in the absence of a g.disclination density, the jump of the inverse deformation field should be characterized by the Burgers vector of the given dislocation density field.

7.1 Derivation of \tilde{Y} in g.disclination theory

Let Π and α be localized in a core as shown in the Figure 21. In g.disclination theory,

$$egin{aligned} & m{Y} = m{S} + \operatorname{grad} m{W} \ & m{lpha} & := m{S} : m{X} + \operatorname{grad} m{W} : m{X} \end{aligned}$$

with

$$\operatorname{curl} \boldsymbol{Y} = \operatorname{curl} \boldsymbol{S} = \boldsymbol{\Pi}.$$

From the definition of $\boldsymbol{\alpha}$, we have

$$\alpha_{ij} = S_{imn}e_{jmn} + W_{im,n}e_{jmn}$$

$$\Rightarrow \alpha_{ij}e_{jrs} = (S_{imn} + W_{im,n}) \left(\delta_{mr}\delta_{ns} - \delta_{ms}\delta_{nr}\right)$$

$$\Rightarrow \left(Y_{irs} - \frac{1}{2}\alpha_{ij}e_{jrs}\right) - \left(Y_{isr} - \frac{1}{2}\alpha_{ij}e_{jsr}\right) = 0$$

Now, we define \tilde{Y} , in terms of ingredients of g.disclination theory, as

$$\tilde{Y}_{imn} := Y_{imn} - \frac{1}{2}\alpha_{ij}e_{jmn} = S_{imn} + W_{im,n} - \frac{1}{2}\alpha_{ij}e_{jmn}$$
(20)

and verify that

$$\tilde{Y}_{imn} = \tilde{Y}_{inm}$$

Therefore, \tilde{Y} is symmetric in the last two indices. In addition,

$$(\operatorname{curl} \tilde{\boldsymbol{Y}})_{imr} = e_{rpn} \tilde{Y}_{imn,p} = S_{imn,p} e_{rpn} - W_{im,np} e_{rpn} - \frac{1}{2} \alpha_{il,p} e_{lmn} e_{rpn}$$
$$\Rightarrow (\operatorname{curl} \tilde{\boldsymbol{Y}})_{imr} = \Pi_{imr} - \frac{1}{2} \alpha_{il,p} e_{lmn} e_{rpn}.$$

Since Π and α are localized in the core, curl \tilde{Y} is localized in the core and curl $\tilde{Y} = 0$ outside the core.

7.2 Jump of inverse deformation in terms of defect strengths in g.disclination theory

With the definition of Y in terms of g.disclination theory (20), we will now identify Δ and the jump of the inverse deformation across a cut-surface in a canonical example in terms of prescribed data used to define an isolated defect line (i.e. the strengths of the g.disclination and dislocation contained in it) and location on the surface.

A g.disclination in a thick infinite plate in the $x_1 - x_2$ plane is considered, with the g.disclination line in the positive x_3 direction. Assume the strength of the g.disclination to be Δ^F (cf. Sec. 6 for definition of the strength). Based on the characterization of $\boldsymbol{\Pi}$ in (10), a candidate for the localized and smooth generalized disclination density $\boldsymbol{\Pi}$ is assumed to only have non-zero components Π_{ij3} , namely $\boldsymbol{\Pi} = \Pi_{ij3} \boldsymbol{e}_i \otimes \boldsymbol{e}_j \otimes \boldsymbol{e}_3$, with Π_{ij3} given as (cf. [Ach01])

$$\Pi_{ij3} = \psi_{ij}(r) = \begin{cases} \frac{\Delta_{ij}^F}{\pi r_0} \left(\frac{1}{r} - \frac{1}{r_0}\right) & r < r_0 \\ 0 & r \ge r_0, \end{cases}$$

where $r = \sqrt{x_1^2 + x_2^2}$. It is easy to verify that $\boldsymbol{\Pi}$ is a smooth field and

$$\int_{core} \boldsymbol{\Pi} \boldsymbol{e}_3 d\boldsymbol{a} = \boldsymbol{\Delta}^F.$$

Similarly, the localized and smooth dislocation density α is assumed as

$$\alpha_{i3} = \begin{cases} \frac{b_i}{\pi r_0} \left(\frac{1}{r} - \frac{1}{r_0}\right) & r < r_0\\ 0 & r \ge r_0, \end{cases}$$

where \boldsymbol{b} is the Burgers vector with

$$\int_{core} \alpha \boldsymbol{e}_3 d\boldsymbol{a} = \boldsymbol{b}.$$

Recall that

$$\tilde{Y}_{irs} = Y_{irs} - \frac{1}{2} \alpha_{ij} e_{rsj}$$

$$\Rightarrow \tilde{Y}_{irs} = S_{irs}^{\perp} + Z_{ir,s}^{s} + W_{ir,s} - \frac{1}{2} \alpha_{ij} e_{rsj}$$

$$\Rightarrow \tilde{Y} = S^{\perp} + \text{grad} Z^{s} + \text{grad} W - \frac{1}{2} \alpha : X,$$
(21)

where Z^s is the compatible part of S and S^{\perp} is incompatible part of S that cannot be represented as the gradient, satisfying the equations

$$\operatorname{curl} S^{\perp} = \Pi$$

 $\operatorname{div} S^{\perp} = 0$
 $S^{\perp} n = 0$ on the boundary.

One way to get the solution of S^{\perp} from the above equations is to decompose S^{\perp} as $S^{\perp} = S^* + \text{grad } Z^*$, where S^* satisfies

$$\operatorname{curl} \boldsymbol{S}^* = \boldsymbol{\Pi}$$

div $\boldsymbol{S}^* = \boldsymbol{0}$ (22)

and Z^* satisfies

$$\operatorname{div}(\operatorname{grad} \boldsymbol{Z}^*) = \boldsymbol{0}$$

$$(\operatorname{grad} \boldsymbol{Z}^*)\boldsymbol{n} = -\boldsymbol{S}^*\boldsymbol{n} \quad \text{on the boundary.}$$
(23)

The solution of (22) can be acquired from the Riemann-Graves operator as shown in Appendix A. Furthermore, since $\int_{\partial V} \mathbf{S}^* \mathbf{n} da = \int_V \operatorname{div} \mathbf{S}^* dv = \mathbf{0}$, where ∂V is the boundary of V, a unique solution for grad \mathbf{Z}^* from (23), which is the (component-wise) Laplace equation for \mathbf{Z}^* with Neumann boundary conditions, exists. Substituting \mathbf{S}^{\perp} into Eqn (21), we have

$$\tilde{\mathbf{Y}} = \mathbf{S}^{\perp} + \operatorname{grad} \mathbf{Z}^{s} + \operatorname{grad} \mathbf{W} - \frac{1}{2}\boldsymbol{\alpha} : \mathbf{X}$$

$$\Rightarrow \tilde{\mathbf{Y}} = \mathbf{S}^{*} + \operatorname{grad} \mathbf{Z}^{*} + \operatorname{grad} \mathbf{Z}^{s} + \operatorname{grad} \mathbf{W} - \frac{1}{2}\boldsymbol{\alpha} : \mathbf{X}$$

$$\Rightarrow \tilde{\mathbf{Y}} := \mathbf{S}^{*} + \operatorname{grad} \mathbf{A} - \frac{1}{2}\boldsymbol{\alpha} : \mathbf{X},$$
(24)

where A is defined as $A := Z^* + Z^s + W$.

In addition, we have

$$\alpha_{ij} = S_{imn}e_{jmn} + W_{im,n}e_{jmn}$$

$$\alpha_{ij} = S^*_{imn}e_{jmn} + Z^*_{im,n}e_{jmn} + Z^s_{im,n}e_{jmn} + W_{im,n}e_{jmn}$$

$$\alpha_{ij} - S^*_{imn}e_{jmn} = -(\operatorname{curl} \boldsymbol{A})_{ij}.$$
(25)

Denoting $B_{ij} = S_{imn}^* e_{jmn} - \alpha_{ij}$, we have

$$(\operatorname{curl} \boldsymbol{A})_{ij} = B_{ij}.$$

As given in Appendix A, we obtain S^* as

$$S_{ij1}^{*} = \begin{cases} \frac{\Delta_{ij}^{F}}{2\pi} \left(-\frac{x_{2}}{r^{2}}\right) & r > r_{0} \\ \frac{-x_{2}\Delta_{ij}^{F}}{\pi r^{2}r_{0}} \left(r - \frac{r^{2}}{2r_{0}}\right) & r \le r_{0} \end{cases}$$
$$S_{ij2}^{*} = \begin{cases} \frac{\Delta_{ij}^{F}}{2\pi} \left(\frac{x_{1}}{r^{2}}\right) & r > r_{0} \\ \frac{x_{1}\Delta_{ij}^{F}}{\pi r^{2}r_{0}} \left(r - \frac{r^{2}}{2r_{0}}\right) & r \le r_{0}. \end{cases}$$

Also, following similar arguments as in Appendix A, in Appendix B we obtain A^* with $A = A^* + \operatorname{grad} z^A$,⁴ where A^* is given by

$$\begin{aligned} A_{11}^{*} &= \begin{cases} C_{1} \left(-\Delta_{12}^{F} x_{2}^{2} - \Delta_{11}^{F} x_{1} x_{2} \right) + \frac{x_{2}}{r^{2}} \left[\frac{b_{1}}{\pi r_{0}} \left(r - \frac{r^{2}}{2r_{0}} \right) \right] & r < r_{0} \\ C_{2} \left(-\Delta_{12}^{F} x_{2}^{2} - \Delta_{11}^{F} x_{1} x_{2} \right) + \frac{x_{2}}{r^{2}} \frac{b_{1}}{2\pi} & r \ge r_{0} \end{cases} \\ A_{12}^{*} &= \begin{cases} C_{1} \left(\Delta_{12}^{F} x_{2} x_{1} + \Delta_{11}^{F} x_{1}^{2} \right) - \frac{x_{1}}{r^{2}} \left[\frac{b_{1}}{\pi r_{0}} \left(r - \frac{r^{2}}{2r_{0}} \right) \right] & r < r_{0} \\ C_{2} \left(\Delta_{12}^{F} x_{2} x_{1} + \Delta_{11}^{F} x_{1}^{2} \right) - \frac{x_{1}}{r^{2}} \frac{b_{1}}{2\pi} & r \ge r_{0} \end{cases} \\ A_{21}^{*} &= \begin{cases} C_{1} \left(-\Delta_{22}^{F} x_{2}^{2} - \Delta_{21}^{F} x_{1} x_{2} \right) + \frac{x_{2}}{r^{2}} \left[\frac{b_{2}}{\pi r_{0}} \left(r - \frac{r^{2}}{2r_{0}} \right) \right] & r < r_{0} \\ C_{2} \left(-\Delta_{22}^{F} x_{2}^{2} - \Delta_{21}^{F} x_{1} x_{2} \right) + \frac{x_{2}}{r^{2}} \frac{b_{2}}{2\pi} & r \ge r_{0} \end{cases} \\ A_{21}^{*} &= \begin{cases} C_{1} \left(\Delta_{22}^{F} x_{2} x_{1} + \Delta_{21}^{F} x_{1}^{2} \right) - \frac{x_{1}}{r^{2}} \left[\frac{b_{2}}{\pi r_{0}} \left(r - \frac{r^{2}}{2r_{0}} \right) \right] & r < r_{0} \\ C_{2} \left(\Delta_{22}^{F} x_{2} x_{1} + \Delta_{21}^{F} x_{1}^{2} \right) - \frac{x_{1}}{r^{2}} \left[\frac{b_{2}}{\pi r_{0}} \left(r - \frac{r^{2}}{2r_{0}} \right) \right] & r < r_{0} \\ C_{2} \left(\Delta_{22}^{F} x_{2} x_{1} + \Delta_{21}^{F} x_{1}^{2} \right) - \frac{x_{1}}{r^{2}} \frac{b_{2}}{2\pi} & r \ge r_{0} \end{cases} \end{cases} \end{cases}$$

and $C_1 = \frac{1}{2\pi r_0 r} - \frac{1}{6\pi r_0^2}$ and $C_2 = \frac{1}{3\pi r_0 r} + \frac{r-r_0}{2\pi r^3}$. A^* can be decomposed into two parts. The first part is the terms associated with C_1 and C_2 , denoted as A^o ; the other part is the remaining terms associated with b, denoted as A^{α} . Thus, $A = A^o + A^{\alpha} + \text{grad } z^A$. A^o and A^{α} are given as

$$\begin{split} A_{11}^{o} &= \begin{cases} \left(\frac{1}{2\pi r_{0}r} - \frac{1}{6\pi r_{0}^{2}}\right) \left(-\Delta_{12}^{F} x_{2}^{2} - \Delta_{11}^{F} x_{1} x_{2}\right) & r < r_{0} \\ \left(\frac{1}{3\pi r_{0}r} + \frac{r - r_{0}}{2\pi r^{3}}\right) \left(-\Delta_{12}^{F} x_{2}^{2} - \Delta_{11}^{F} x_{1} x_{2}\right) & r \geq r_{0} \end{cases} \\ A_{12}^{o} &= \begin{cases} \left(\frac{1}{2\pi r_{0}r} - \frac{1}{6\pi r_{0}^{2}}\right) \left(\Delta_{12}^{F} x_{2} x_{1} + \Delta_{11}^{F} x_{1}^{2}\right) & r < r_{0} \\ \left(\frac{1}{3\pi r_{0}r} + \frac{r - r_{0}}{2\pi r^{3}}\right) \left(\Delta_{12}^{F} x_{2} x_{1} + \Delta_{11}^{F} x_{1}^{2}\right) & r \geq r_{0} \end{cases} \\ A_{21}^{o} &= \begin{cases} \left(\frac{1}{2\pi r_{0}r} - \frac{1}{6\pi r_{0}^{2}}\right) \left(-\Delta_{22}^{F} x_{2}^{2} - \Delta_{21}^{F} x_{1} x_{2}\right) & r < r_{0} \\ \left(\frac{1}{3\pi r_{0}r} + \frac{r - r_{0}}{2\pi r^{3}}\right) \left(-\Delta_{22}^{F} x_{2}^{2} - \Delta_{21}^{F} x_{1} x_{2}\right) & r \geq r_{0} \end{cases} \\ A_{21}^{o} &= \begin{cases} \left(\frac{1}{2\pi r_{0}r} - \frac{1}{6\pi r_{0}^{2}}\right) \left(\Delta_{22}^{F} x_{2} x_{1} + \Delta_{21}^{F} x_{1}^{2}\right) & r < r_{0} \\ \left(\frac{1}{3\pi r_{0}r} + \frac{r - r_{0}}{2\pi r^{3}}\right) \left(\Delta_{22}^{F} x_{2} x_{1} + \Delta_{21}^{F} x_{1}^{2}\right) & r < r_{0} \end{cases} \\ A_{22}^{o} &= \begin{cases} \left(\frac{1}{2\pi r_{0}r} - \frac{1}{6\pi r_{0}^{2}}\right) \left(\Delta_{22}^{F} x_{2} x_{1} + \Delta_{21}^{F} x_{1}^{2}\right) & r < r_{0} \\ \left(\frac{1}{3\pi r_{0}r} + \frac{r - r_{0}}{2\pi r^{3}}\right) \left(\Delta_{22}^{F} x_{2} x_{1} + \Delta_{21}^{F} x_{1}^{2}\right) & r < r_{0} \end{cases} \\ A_{22}^{o} &= \begin{cases} \left(\frac{1}{2\pi r_{0}r} - \frac{1}{6\pi r_{0}^{2}}\right) \left(\Delta_{22}^{F} x_{2} x_{1} + \Delta_{21}^{F} x_{1}^{2}\right) & r < r_{0} \\ \left(\frac{1}{3\pi r_{0}r} + \frac{r - r_{0}}{2\pi r^{3}}\right) \left(\Delta_{22}^{F} x_{2} x_{1} + \Delta_{21}^{F} x_{1}^{2}\right) & r < r_{0} \end{cases} \\ A_{11}^{\alpha} &= \begin{cases} \frac{x_{2}}{r^{2}} \left[\frac{b_{1}}{\pi r_{0}} \left(r - \frac{r^{2}}{2r_{0}}\right)\right] & r < r_{0} \\ \frac{x_{2}}{r^{2}} \frac{b_{2}}{2\pi} & r \geq r_{0} \end{cases} \\ A_{21}^{\alpha} &= \begin{cases} \frac{x_{1}}{r^{2}} \left[\frac{b_{2}}{\pi r_{0}} \left(r - \frac{r^{2}}{2r_{0}}\right)\right] & r < r_{0} \\ \frac{x_{1}}{r^{2}} \frac{b_{2}}}{2\pi} & r \geq r_{0} \end{cases} \\ A_{22}^{\alpha} &= \begin{cases} \frac{x_{1}}{r^{2}} \left[\frac{b_{2}}{\pi r_{0}} \left(r - \frac{r^{2}}{2r_{0}}\right)\right] & r < r_{0} \\ \frac{x_{1}}{r^{2}} \frac{b_{2}}}{2\pi} & r \geq r_{0}. \end{cases} \end{cases} \end{cases}$$

We now consider a multiply connected domain by thinking of the cylinder with the core region excluded and introduce a simply-connected domain by a cut-surface. On this simply-connected

⁴While not relevant for the essentially topological arguments here, we note that it is in the field $gradz^A$ that the compatible part of W resides which helps in satisfaction of force equilibrium.

domain we define a \tilde{W} satisfying grad $\tilde{W} = \tilde{Y}$ and recall that $\tilde{Y} = S^* + \operatorname{grad} A - \frac{1}{2}\alpha : X$. Then

$$\tilde{\boldsymbol{W}}(\boldsymbol{x}) = \int_{\boldsymbol{x}_r}^{\boldsymbol{x}} \tilde{\boldsymbol{Y}}(\boldsymbol{s}) d\boldsymbol{s} + \tilde{\boldsymbol{W}}(\boldsymbol{x}_r)$$
(26)

$$\tilde{\boldsymbol{W}}(\boldsymbol{x}) = \int_{\boldsymbol{x}_r}^{\boldsymbol{x}} [\boldsymbol{S}^* + \operatorname{grad} \boldsymbol{A} - \frac{1}{2}\boldsymbol{\alpha} : \boldsymbol{X}](\boldsymbol{s})d\boldsymbol{s} + \tilde{\boldsymbol{W}}(\boldsymbol{x}_r)$$
(27)

$$\tilde{\boldsymbol{W}}(\boldsymbol{x}) = \int_{\boldsymbol{x}_r}^{\boldsymbol{x}} \boldsymbol{S}^*(\boldsymbol{s}) d\boldsymbol{s} - \frac{1}{2} \int_{\boldsymbol{x}_r}^{\boldsymbol{x}} (\boldsymbol{\alpha} : \boldsymbol{X})(\boldsymbol{s}) d\boldsymbol{s} + \boldsymbol{A}(\boldsymbol{x}) + const,$$
(28)

where \boldsymbol{x}_r is a given point and *const* is the constant $\tilde{\boldsymbol{W}}(\boldsymbol{x}_r) - \boldsymbol{A}(\boldsymbol{x}_r)$, with $\tilde{\boldsymbol{W}}(\boldsymbol{x}_r)$ being arbitrarily assignable.

Consider a path p from z^- to z^+ , both points arbitrarily close to $z \in S$, on opposite sides of S (see (14) for notation) and define

$$\llbracket oldsymbol{y}(oldsymbol{z})
rbracket = \lim_{\substack{oldsymbol{z}^+ o oldsymbol{z} \ oldsymbol{z}^- o oldsymbol{z}}} \int_p ilde{oldsymbol{W}}(oldsymbol{x}) doldsymbol{x}$$

for any z on the cut-surface.

After substituting \tilde{W} and noticing $\oint_p const dx = 0$ and $\oint_p \operatorname{grad} z^A dx = 0$, the jump at x can be further written as

$$\llbracket \boldsymbol{y} \rrbracket = \int_{p} \int_{\boldsymbol{x}_{r}}^{\boldsymbol{x}} \boldsymbol{S}^{*}(\boldsymbol{s}) d\boldsymbol{s} d\boldsymbol{x} + \int_{p} \boldsymbol{A}(\boldsymbol{x}) d\boldsymbol{x} - \frac{1}{2} \int_{p} \int_{\boldsymbol{x}_{r}}^{\boldsymbol{x}} (\boldsymbol{\alpha} : \boldsymbol{X})(\boldsymbol{s}) d\boldsymbol{s} d\boldsymbol{x}$$
$$= \int_{p} \int_{\boldsymbol{x}_{r}}^{\boldsymbol{x}} \boldsymbol{S}^{*}(\boldsymbol{s}) d\boldsymbol{s} d\boldsymbol{x} + \int_{p} \boldsymbol{A}^{o}(\boldsymbol{x}) d\boldsymbol{x} + \int_{p} \boldsymbol{A}^{\alpha}(\boldsymbol{x}) d\boldsymbol{x} - \frac{1}{2} \int_{p} \int_{\boldsymbol{x}_{r}}^{\boldsymbol{x}} (\boldsymbol{\alpha} : \boldsymbol{X})(\boldsymbol{s}) d\boldsymbol{s} d\boldsymbol{x}.$$
(29)

Now suppose \boldsymbol{x} located as (R, 0) and p is given as a circle with radius r = R, where $R \ge r_0$. Clearly, this circle encloses the whole disclination core. Also, since $A_{ij}^o(x_1, x_2) = A_{ij}^o(-x_1, -x_2)$, then

$$\int_p oldsymbol{A}^o(oldsymbol{x}) doldsymbol{x} = oldsymbol{0}.$$

Also, for any loop enclosing the core, $\alpha = 0$ along the loop and thus,

$$\int_p \int_{\boldsymbol{x}_r}^{\boldsymbol{x}} (\boldsymbol{\alpha}: \boldsymbol{X})(\boldsymbol{s}) d\boldsymbol{s} d\boldsymbol{x} = \boldsymbol{0}$$

Therefore, the jump $\llbracket \boldsymbol{y} \rrbracket = \llbracket \boldsymbol{y} \rrbracket^s + \llbracket \boldsymbol{y} \rrbracket^{\alpha}$, where $\llbracket \boldsymbol{y} \rrbracket^s = \oint_p \int_{\boldsymbol{x}_r}^{\boldsymbol{x}} \boldsymbol{S}^*(\boldsymbol{s}) d\boldsymbol{s} d\boldsymbol{x}$ and $\llbracket \boldsymbol{y} \rrbracket^{\alpha} = \oint_p \boldsymbol{A}^{\alpha}(\boldsymbol{x}) d\boldsymbol{x}$. With reference to Figure 22 and choose \boldsymbol{x}_r as \boldsymbol{x}^- , $\llbracket \boldsymbol{y} \rrbracket^s$ evaluates to

$$\begin{bmatrix} \boldsymbol{y} \end{bmatrix}_{1}^{s} = R \Delta_{11}^{F}$$

$$\llbracket \boldsymbol{y} \rrbracket_{2}^{s} = R \Delta_{21}^{F}.$$
(30)

 $\llbracket \boldsymbol{y} \rrbracket^{\alpha} = \oint_p \boldsymbol{A}^{\alpha}(\boldsymbol{x}) d\boldsymbol{x}$ can be obtained as follows:

$$\begin{split} \llbracket \boldsymbol{y} \rrbracket_{1}^{\alpha} &= \oint_{2\pi}^{0} \left[-A_{11}^{\alpha} R \sin \beta + A_{12}^{\alpha} R(\cos \beta) \right] d\beta \\ &\Rightarrow \llbracket \boldsymbol{y} \rrbracket_{1}^{\alpha} = \oint_{2\pi}^{0} -\frac{b_{1}}{2\pi} d\beta = b_{1} \\ \llbracket \boldsymbol{y} \rrbracket_{2}^{\alpha} &= \oint_{2\pi}^{0} \left[-A_{21}^{\alpha} R \sin \beta + A_{22}^{\alpha} R(\cos \beta) \right] d\beta \\ &\Rightarrow \llbracket \boldsymbol{y} \rrbracket_{2}^{\alpha} = \oint_{2\pi}^{0} -\frac{b_{2}}{2\pi} d\beta = b_{2} \end{split}$$



Figure 22: Configuration of a closed contour enclosing the disclination and passing through x.

Thus, the jump at point \boldsymbol{x} , (R, 0), is

$$\llbracket \boldsymbol{y} \rrbracket_1 = R \Delta_{11}^F + b_1$$
$$\llbracket \boldsymbol{y} \rrbracket_2 = R \Delta_{21}^F + b_2,$$

which can be written in the form

$$\llbracket \boldsymbol{y} \rrbracket(\boldsymbol{x}) = \boldsymbol{\Delta}^{F} \boldsymbol{x} + \boldsymbol{b} = \boldsymbol{\Delta}^{F} (\boldsymbol{x} - \boldsymbol{x}_{0}) + \llbracket \boldsymbol{y} \rrbracket(\boldsymbol{x}_{0}),$$
(31)

where

$$\boldsymbol{\Delta}^{F} = \begin{bmatrix} \Delta_{11}^{F} & \Delta_{12}^{F} \\ \Delta_{21}^{F} & \Delta_{22}^{F} \end{bmatrix}$$

and

$$\llbracket \boldsymbol{y} \rrbracket(\boldsymbol{x}_0) = \boldsymbol{\Delta}^F \boldsymbol{x}_0 + \boldsymbol{b}.$$

for an arbitrarily chosen base-point x_0 on the cut-surface.

For $\Pi = \mathbf{0}$ (i.e. no generalized disclination in the defect), given a localized dislocation density $\boldsymbol{\alpha}$, the jump in the inverse deformation should be the same as the integral of $\boldsymbol{\alpha}$ over any arbitrary area threaded by the core, denoted as **b**. Since $\Pi = \mathbf{0}$, then $\Delta^F = \mathbf{0}$ and (31) implies,

$$\llbracket y
rbracket = b.$$

Thus, in this special, but canonical, example, we have characterized the jump in the inverse deformation due to a defect line in terms of data characterizing the g.disclination and dislocation densities of g.disclination theory and shown that the result is consistent with what is expected in the simpler case when the g.disclination density vanishes.

7.3 The connection between W and y

We now deal with the question of how the inverse deformation field \boldsymbol{y} defined on a cut-surface induced simply-connected domain defined from the field $\tilde{\boldsymbol{Y}}$ may be related to the i-elastic 1-distortion field \boldsymbol{W} of g.disclination theory. The setting we have in mind is as follows: with reference to Fig.



Figure 23: The cross-section of a simply-connected domain Ω with a cut-surface S and a core Ω_c . The blue shaded area is the layer S_l which includes the core.

23, we consider the domain Ω with the core comprising the region $\Omega_c \subset \Omega$. Let the cut-surface be S, connecting a curve on the boundary of Ω_c to a curve on the boundary of Ω so that $(\Omega \setminus \Omega_c) \setminus S$ is simply-connected. Also consider a 'layer' region $S_l \subset \Omega$ such that $S \subset S_l$ as well as $\Omega_c \subset S_l$, as shown in Figure 23. We assume that S has support in S_l . We now think that a problem of g.disclination theory has been solved with S, Π , and α as given data on Ω satisfying the constraint curl $S = \Pi$.

From (20) and (26), we have

grad
$$(\tilde{\boldsymbol{W}} - \boldsymbol{W}) = \boldsymbol{S} - \frac{1}{2} \boldsymbol{\alpha} \boldsymbol{X}$$
 on $(\boldsymbol{\Omega} \backslash \boldsymbol{\Omega}_c) \backslash \boldsymbol{S}$.

We then have

$$\llbracket \boldsymbol{W} \rrbracket = \llbracket \operatorname{grad} \boldsymbol{y} \rrbracket - \boldsymbol{\Delta}^F = \boldsymbol{0} \quad \text{ on } \mathbf{S}_{\mathbf{y}}$$

since the dislocation density α is localized in the core, an expected result since W is a continuous field on Ω . Moreover, we have

$$\boldsymbol{W} = \operatorname{grad} \boldsymbol{y} + \boldsymbol{W}^* \quad \text{on } \Omega \setminus S_l,$$

where W^* is a constant second-order tensor. Thus, when S_l is truly in the shape of a layer around S (including the core as defined), W can indeed be viewed as the gradient of the deformation y constructed from \tilde{Y} , up to a constant second-order tensor, in most of the domain. On the other hand, when $S_l = \Omega \setminus \Omega_c$ as e.g. when curl $S = \Pi$ with div S = 0 on Ω with Π still supported in the core, such an identification is not possible.

8 Burgers vector of a g.disclination dipole

We will derive the Burgers vector for a given g.disclination dipole with separation vector d. In Figure 24, the red circle is the (cross-section of the) positive g.disclination while the blue circle is

that of the negative g.disclination. The separation vector between these two disclination is d. For the calculations in this section, we assume Cartesian coordinates whose origin is at the positive g.disclination core and the x and y axes are shown as in Figure 24. The boundary of the positive g.disclination core is the circle with center at (0,0) and radius r_0 ; the boundary of the negative disclination core is the circle with center at (d,0) and radius r_0 . Denote the strength of the positive disclination as Δ^F (see Sec. 6 for the definition of the strength). We denote the whole domain as Ω and the boundary of the domain as $\partial\Omega$. The positive g.disclination density field Π^+ and the negative g.disclination density field Π^- are both localized inside the cores (while being defined in all of Ω). We define the core of the g.disclination dipole, Ω_c , as a patch including the positive and negative g.disclinations enclosed by the black contour C in Figure 24. In the following calculation, the core is referred as the g.disclination dipole core. Also, let the cut-surface be S, which is along the positive x axis connecting the boundary of Ω to the curve C at x_c shown as the green line in Figure 24. Namely, $S = \{(x, y) \in \Omega | y = 0, x \ge x_c\}$. In addition, the cut-induced simply-connected domain is denoted as $(\Omega \setminus \Omega_c) \setminus S$.

We denote the defect density field for the g.disclination dipole as $\boldsymbol{\Pi}$. Clearly, $\boldsymbol{\Pi}$ is localized within the g.disclination core, given as $\boldsymbol{\Pi} = \boldsymbol{\Pi}^+ + \boldsymbol{\Pi}^-$ on $\boldsymbol{\Omega}$. Based on the Weingarten-gd theorem, given \boldsymbol{S}^* , $\boldsymbol{\alpha}$, and \boldsymbol{A} , $\tilde{\boldsymbol{Y}}$ is defined as in (21) and (24):

$$\tilde{Y} := S^* + \operatorname{grad} A - \frac{1}{2} \alpha : X$$
 in Ω .

In this case, $\alpha = 0$, thus

 $\tilde{\boldsymbol{Y}} = \boldsymbol{S}^* + \operatorname{grad} \boldsymbol{A}.$ (32)

Recall that given the g.disclination density $\boldsymbol{\Pi}, \boldsymbol{S}^*$ is calculated from

$$\operatorname{curl} {old S}^* = {oldsymbol \Pi} = {oldsymbol \Pi}^+ + {oldsymbol \Pi} - \ \operatorname{div} {old S}^* = {old 0}
ight\} \hspace{1.5cm} ext{in } arOmega$$

Since S^* is a solution to linear equations, S^* can be written as

$$m{S}^{*} = m{S}^{*+} + m{S}^{*-}$$

with S^{*+} and S^{*-} calculated from

$$\begin{array}{l} \operatorname{curl} \boldsymbol{S}^{*+} = \boldsymbol{\Pi}^+ \\ \operatorname{div} \boldsymbol{S}^{*+} = \boldsymbol{0} \end{array} & \text{ in } \boldsymbol{\Omega} \\ \operatorname{curl} \boldsymbol{S}^{*-} = \boldsymbol{\Pi}^- \\ \operatorname{div} \boldsymbol{S}^{*-} = \boldsymbol{0} \end{array} & \text{ in } \boldsymbol{\Omega}. \end{array}$$

Similarly, given S^* , $A = A^* + \operatorname{grad} z^A$, where A^* satisfies (25)

$$\operatorname{curl} \boldsymbol{A}^* = \boldsymbol{S}^* : \boldsymbol{X}$$

 $\operatorname{div} \boldsymbol{A}^* = \boldsymbol{0}$ in $\boldsymbol{\Omega}$.

Therefore, A can be written as $A = A^{*+} + A^{*-} + \text{grad } z^A$, where A^{*+} and A^{*-} are calculated from

$$\begin{array}{l} \operatorname{curl} \boldsymbol{A}^{*+} = \boldsymbol{S}^{*+} : \boldsymbol{X} \\ \operatorname{div} \boldsymbol{A}^{*+} = \boldsymbol{0} \end{array} & \text{ in } \boldsymbol{\Omega} \\ \operatorname{curl} \boldsymbol{A}^{*-} = \boldsymbol{S}^{*-} : \boldsymbol{X} \\ \operatorname{div} \boldsymbol{A}^{*-} = \boldsymbol{0} \end{array} & \text{ in } \boldsymbol{\Omega}. \end{array}$$

After substituting S^* and A, \tilde{Y} can be written as

$$\tilde{Y} = S^{*+} + S^{*-} + \operatorname{grad} A^{*+} + \operatorname{grad} A^{*-} + \operatorname{grad} \operatorname{grad} z^A$$
 on Ω .

In the cut-induced simply-connected domain $(\Omega \setminus \Omega_c) \setminus S$, given the core Ω_c and cut-surface S shown in Figure 24, \tilde{W} is defined as (26)

$$ilde{oldsymbol{W}}(oldsymbol{x}) := \int_{oldsymbol{x}_r}^{oldsymbol{x}} ilde{oldsymbol{Y}}(oldsymbol{s}) doldsymbol{s} + ilde{oldsymbol{W}}(oldsymbol{x}_r)$$

where \boldsymbol{x}_r is a fixed point and $\tilde{\boldsymbol{W}}(\boldsymbol{x}_r)$ is an arbitrary constant. After substituting $\tilde{\boldsymbol{Y}}$, we have

$$\tilde{W}(\boldsymbol{x}) = \int_{\boldsymbol{x}_r}^{\boldsymbol{x}} \boldsymbol{S}^{*+}(\boldsymbol{s}) d\boldsymbol{s} + \int_{\boldsymbol{x}_r}^{\boldsymbol{x}} \boldsymbol{S}^{*-}(\boldsymbol{s}) d\boldsymbol{s} + \boldsymbol{A}^{*+}(\boldsymbol{x}) + \boldsymbol{A}^{*-}(\boldsymbol{x}) + \text{grad}\,\boldsymbol{z}^A(\boldsymbol{x}) + \text{const},$$

where const is the constant equal to $\tilde{W}(\boldsymbol{x}_r) - \boldsymbol{A}^{*+}(\boldsymbol{x}_r) - \boldsymbol{A}^{*-}(\boldsymbol{x}_r) - \operatorname{grad} \boldsymbol{z}^A(\boldsymbol{x}_r)$. Write $\tilde{W}(\boldsymbol{x})$ as

$$\tilde{W}(\boldsymbol{x}) = \tilde{T}^+(\boldsymbol{x};\boldsymbol{x}_r) + \tilde{T}^-(\boldsymbol{x};\boldsymbol{x}_r) + \operatorname{grad} \boldsymbol{z}^A(\boldsymbol{x}) + const,$$

where

With reference to Fig. 24, it follows from (15) that the jump of \boldsymbol{y} at \boldsymbol{x}_0 is

$$\llbracket oldsymbol{y}
rbracket(oldsymbol{x}_0) = \int_p ilde{oldsymbol{W}}(oldsymbol{x}) doldsymbol{x},$$

where p is a path shown in Fig. 24 from \boldsymbol{x}_0^- to \boldsymbol{x}_0^+ . Since z is continuous on Ω , $\int_p \operatorname{grad} \boldsymbol{z}(\boldsymbol{x}) d\boldsymbol{x} = \boldsymbol{0}$. Also, with $\int_p \operatorname{constd} \boldsymbol{x} = \boldsymbol{0}$, we have

$$\llbracket \boldsymbol{y} \rrbracket(\boldsymbol{x}_0) = \int_p \tilde{\boldsymbol{T}}^+(\boldsymbol{x};\boldsymbol{x}_r) d\boldsymbol{x} + \int_p \tilde{\boldsymbol{T}}^-(\boldsymbol{x};\boldsymbol{x}_r) d\boldsymbol{x},$$

Let \mathbf{x}_0 be the point located at $(x_0, 0)$ and denote p^+ as a clockwise circle centered at (0, 0) with radius x_0 and p^- as a clockwise circle centered at (d, 0) with radius $x_0 - d$. Also, we choose \mathbf{x}_0 big enough so that p^+ and p^- enclose the g.disclination core - this induces no loss in generality for our final result, as we show in the discussion surrounding (33) and (34). Also, let \mathbf{x}_r be \mathbf{x}_0^- . Based on the argument in Appendix C,

$$\oint_p ilde{oldsymbol{T}^+}(oldsymbol{x};oldsymbol{x}_r)doldsymbol{x} = \oint_{p^+} ilde{oldsymbol{T}^+}(oldsymbol{x};oldsymbol{x}_0^-)doldsymbol{x} =: oldsymbol{I}^+ \ \oint_p ilde{oldsymbol{T}^-}(oldsymbol{x};oldsymbol{x}_r)doldsymbol{x} = \oint_{p^-} ilde{oldsymbol{T}^-}(oldsymbol{x};oldsymbol{x}_0^-)doldsymbol{x} =: oldsymbol{I}^-$$

(This step is utilized to facilitate the computation of the line integrals, which are most conveniently calculated on circular paths).



Figure 24: Configuration for a disclination dipole with separation vector d and an introduced cut-surface.

Given the Π^+ and Π^- fields, based on the calculation in Appendix D, we have

$$I_1^+ = x_0 \Delta_{11}^F$$

$$I_2^+ = x_0 \Delta_{21}^F$$

$$I_1^- = -(x_0 - d) \Delta_{11}^F$$

$$I_2^- = -(x_0 - d) \Delta_{21}^F.$$

Thus, the jump $\llbracket \boldsymbol{y} \rrbracket$ at \boldsymbol{x}_0 is

$$\llbracket y \rrbracket_1(\boldsymbol{x}_0) = x_0 \Delta_{11}^F - (x_0 - d) \Delta_{11}^F = d\Delta_{11}^F \\ \llbracket y \rrbracket_2(\boldsymbol{x}_0) = x_0 \Delta_{21}^F - (x_0 - d) \Delta_{21}^F = d\Delta_{21}^F$$

Therefore, the jump at \boldsymbol{x}_0 is

$$\llbracket \boldsymbol{y}
rbracket(\boldsymbol{x}_0) = \boldsymbol{\Delta}^F \boldsymbol{d}$$

Recall from (32) that

$$\tilde{Y} = S^* + \operatorname{grad} A,$$

so that

$$\operatorname{curl} \boldsymbol{Y} = \operatorname{curl} \boldsymbol{S}^* = \boldsymbol{\Pi}.$$

Therefore,

$$\boldsymbol{\Delta} = \int_{C} \tilde{\boldsymbol{Y}} d\boldsymbol{x} = \int_{\Omega_{c}} \operatorname{curl} \tilde{\boldsymbol{Y}} \boldsymbol{n} da = \int_{\Omega_{c}} (\boldsymbol{\Pi}^{+} + \boldsymbol{\Pi}^{-}) \boldsymbol{n} da$$

where C is the boundary of the g.disclination core and Ω_c is the area of the core enclosed by C. Because Π^+ and Π^- have opposite signs but are otherwise identical (shifted) fields,

$$\boldsymbol{\Delta} = \int_{\Omega_c} (\boldsymbol{\Pi}^+ + \boldsymbol{\Pi}^-) \boldsymbol{n} d\boldsymbol{a} = \boldsymbol{0}.$$

Then, invoking (17), the jump of \boldsymbol{y} satisfies

$$\llbracket \boldsymbol{y} \rrbracket(\boldsymbol{x}) = \llbracket \boldsymbol{y} \rrbracket(\boldsymbol{x}_0) = \boldsymbol{\Delta}^F \boldsymbol{d} \quad \text{for} \quad \boldsymbol{x} \in S.$$
(33)

In addition, as proved in Section 6.2, the jump $\llbracket y \rrbracket$ is independent of the cut-surface when $\Delta = 0$. Therefore, defining the Burgers vector **b** of the g.disclination dipole as the jump $\llbracket y \rrbracket$ across any arbitrary cut-surface rendering $\Omega \backslash \Omega_c$ simply-connected, we have

$$\boldsymbol{b} = \boldsymbol{\Delta}^F \boldsymbol{d}.\tag{34}$$

Now, considering a disclination dipole with separation vector $\boldsymbol{d} = \delta \boldsymbol{r}$, where the Frank vector for the positive disclination is given as $\boldsymbol{\Omega} = \Omega_3 \boldsymbol{e}_3$, $\boldsymbol{\Delta}^F$ is given by the difference of two inverse rotation tensors, written as

$$\begin{bmatrix} \boldsymbol{\Delta}^F \end{bmatrix} = \begin{bmatrix} \cos \Omega_3 - 1 & -\sin \Omega_3 \\ \sin \Omega_3 & \cos \Omega_3 - 1 \end{bmatrix}.$$

When the strength of the disclination is low, $|\Omega_3| \ll 1$, $\cos \Omega_3 - 1 \approx 0$ and $\sin \Omega_3 \approx \Omega_3$, and we have

$$\boldsymbol{\Delta}^{F} = \begin{bmatrix} 0 & -\Omega_{3} \\ \Omega_{3} & 0 \end{bmatrix},$$

a skew symmetric tensor.

From (34), the components of the Burgers vector **b** are given as

$$b_1 = -\Omega_3 \delta r_2, \quad b_2 = \Omega_3 \delta r_1$$

which matches the result (7) from linear elasticity.

Acknowledgments

CZ and AA acknowledge support from grant NSF-DMS-1434734. AA also acknowledges support from grants NSF-CMMI-1435624 and ARO W911NF-15-1-0239. It is a pleasure to acknowledge a discussion with Jianfeng Lu.

A Analytical solution for S^* in the generalized disclination model

We recall the governing equations of S^* in g.disclination theory given as

$$\operatorname{curl} S^* = \Pi$$

 $\operatorname{div} S^* = 0.$

To solve S^* from these two equations, the Riemann-Graves integral operator [Ede85, EL88, Ach01] is applied. Suppose Π satisfies $\Pi_{ijk,k} = 0$, and define

$$\ddot{\Pi}_{injk} = e_{jkm} \Pi_{inm}.$$

In the 2-D case, the only non-zero components of $\boldsymbol{\Pi}$ are Π_{ij3} with i, j = 1, 2. Thus $\hat{\boldsymbol{\Pi}}$ has no non-zero component with index 3. Then the relation between $\boldsymbol{\Pi}$ and $\hat{\boldsymbol{\Pi}}$ is

$$\begin{aligned} \Pi_{1112} &= \Pi_{113} & \Pi_{2212} &= \Pi_{223} & \Pi_{1212} &= \Pi_{123} & \Pi_{2112} &= \Pi_{213} \\ \hat{\Pi}_{1121} &= -\Pi_{113} & \hat{\Pi}_{2221} &= -\Pi_{223} & \hat{\Pi}_{1221} &= -\Pi_{123} & \hat{\Pi}_{2121} &= -\Pi_{213} \end{aligned}$$

The integral is now introduced as

$$H_{ink} = (x_j - x_j^0) \int_0^1 \hat{H}_{injk}(x^0 + \lambda(x - x^0))\lambda d\lambda,$$

where x^0 is any fixed point in the body (and we assume a star-shaped domain).

Suppose Π_{ij3} takes the form

$$\Pi_{ij3} = \psi_{ij}(\sqrt{x_1^2 + x_2^2}),$$

and also assume x^0 be the origin of the coordinates. Then for a positive disclination, the H at x is given as

$$H_{ij1} = \begin{cases} \frac{\Delta F_{ij}}{2\pi} (-\frac{x_2}{r^2}) & r > r_0 \\ \frac{-x_2}{r^2} \int_0^r \psi_{ij}(s) s ds & r \le r_0 \end{cases}$$
$$H_{ij2} = \begin{cases} \frac{\Delta F_{ij}}{2\pi} (\frac{x_1}{r^2}) & r > r_0 \\ \frac{x_1}{r^1} \int_0^r \psi_{ij}(s) s ds & r \le r_0 \end{cases}$$

Then it can be shown that H is the solution of S^* which satisfies curl $S^* = \Pi$ and div $S^* = 0$. In this work, we assume ψ takes the form

$$\psi_{ij}(r) = \begin{cases} \frac{\Delta F_{ij}}{\pi r_0} (\frac{1}{r} - \frac{1}{r_0}) & r < r_0 \\ 0 & r \ge r_0. \end{cases}$$

Thus,

$$H_{ij1} = \begin{cases} \frac{\Delta F_{ij}}{2\pi} \left(-\frac{x_2}{r^2}\right) & r > r_0\\ \frac{-x_2 \Delta F_{ij}}{\pi r^2 r_0} \left(r - \frac{r^2}{2r_0}\right) & r \le r_0\\ \end{cases}$$
$$H_{ij2} = \begin{cases} \frac{\Delta F_{ij}}{2\pi} \left(\frac{x_1}{r^2}\right) & r > r_0\\ \frac{x_1 \Delta F_{ij}}{\pi r^2 r_0} \left(r - \frac{r^2}{2r_0}\right) & r \le r_0. \end{cases}$$

B Calculation of A^* for \tilde{Y}

The governing equations for A^* are given as

$$\operatorname{curl} \boldsymbol{A}^* = \boldsymbol{S}^* : \boldsymbol{X}$$

 $\operatorname{div} \boldsymbol{A}^* = \boldsymbol{0}$

Since in the 2-D case the last index of all non-zero components of $\boldsymbol{\Pi}$ is 3 and the first two are 1 or 2, $(div (\boldsymbol{S}^* : \boldsymbol{X}))_{ipp} = -\Pi_{ipp} = 0$, indicating solutions to \boldsymbol{A}^* exist and we again utilize the Riemann-Graves integral.

Denote $B = S^* : X$, then $B_{13} = S^*_{112} - S^*_{121}$ and $B_{23} = S^*_{212} - S^*_{221}$. After substituting the S^* from Appendix A,

$$B_{13} = \begin{cases} \frac{1}{2\pi r^2} (\Delta F_{12} x_2 + \Delta F_{11} x_1) & r \ge r_0 \\ \frac{1}{r^2} (x_2 \int_0^r \psi_{12} s ds + x_1 \int_0^r \psi_{11} s ds) & r < r_0 \end{cases}$$
$$B_{23} = \begin{cases} \frac{1}{2\pi r^2} (\Delta F_{22} x_2 + \Delta F_{21} x_1) & r \ge r_0 \\ \frac{1}{r^2} (x_2 \int_0^r \psi_{22} s ds + x_1 \int_0^r \psi_{21} s ds) & r < r_0 \end{cases}$$

In addition, we can assume ψ_{ij} also takes the form (as in Appendix A)

$$\psi_{ij}(r) = \begin{cases} \frac{\Delta F_{ij}}{\pi r_0} (\frac{1}{r} - \frac{1}{r_0}) & r < r_0 \\ 0 & r \ge r_0. \end{cases}$$

Thus,

$$B_{13} = \begin{cases} \frac{1}{2\pi r^2} (\Delta F_{12} x_2 + \Delta F_{11} x_1) & r \ge r_0\\ \frac{1}{\pi r^2 r_0} (r - \frac{r^2}{2r_0}) (\Delta F_{12} x_2 + \Delta F_{11} x_1) & r < r_0 \end{cases}$$

$$B_{23} = \begin{cases} \frac{1}{2\pi r^2} (\Delta F_{22} x_2 + \Delta F_{21} x_1) & r \ge r_0 \\ \frac{1}{\pi r^2 r_0} (r - \frac{r^2}{2r_0}) (\Delta F_{22} x_2 + \Delta F_{21} x_1) & r < r_0 \end{cases}$$

The Riemann-Grave integral operator for the equation $\operatorname{curl} A^* = B$ is

$$A_{ij}^{*} = (x_m - x_m^0) \int_0^1 \hat{B}_{imj}(x^0 + \lambda(x - x^0))\lambda d\lambda,$$

where $\hat{B}_{imj} = e_{mjr}B_{ir}$. Assuming \boldsymbol{x}^0 to be the origin, \boldsymbol{A}^* can be written as

$$\begin{aligned} A_{11}^* &= -x_2 \int_0^1 B_{13}(\lambda x) \lambda d\lambda \\ A_{12}^* &= x_1 \int_0^1 B_{13}(\lambda x) \lambda d\lambda \\ A_{21}^* &= -x_2 \int_0^1 B_{23}(\lambda x) \lambda d\lambda \\ A_{22}^* &= x_1 \int_0^1 B_{23}(\lambda x) \lambda d\lambda. \end{aligned}$$

Also, with the choice of x^0 as the origin, and if $r < r_0$, then $\lambda r < r_0$, and we have

$$\int_{0}^{1} B_{13}(\lambda \boldsymbol{x}) \lambda d\lambda = (1 - \frac{r}{3r_0}) \left(\frac{\Delta F_{12} \sin \theta}{2\pi r_0} + \frac{\Delta F_{11} \cos \theta}{2\pi r_0} \right) \int_{0}^{1} B_{23}(\lambda \boldsymbol{x}) \lambda d\lambda = (1 - \frac{r}{3r_0}) \left(\frac{\Delta F_{22} \sin \theta}{2\pi r_0} + \frac{\Delta F_{21} \cos \theta}{2\pi r_0} \right)$$

If $r \geq r_0$, then

$$\int_0^1 B_{13}(\lambda \boldsymbol{x}) \lambda d\lambda = \int_0^{r_0/r} B_{13}(\lambda \boldsymbol{x}) \lambda d\lambda + \int_{r_0/r}^1 B_{13}(\lambda \boldsymbol{x}) \lambda d\lambda$$
$$\int_0^1 B_{23}(\lambda \boldsymbol{x}) \lambda d\lambda = \int_0^{r_0/r} B_{23}(\lambda \boldsymbol{x}) \lambda d\lambda + \int_{r_0/r}^1 B_{23}(\lambda \boldsymbol{x}) \lambda d\lambda.$$

Since,

$$\int_{r_0/r}^r B_{13}(\lambda \boldsymbol{x}) \lambda d\lambda = \frac{r - r_0}{2\pi r^2} (\Delta F_{12} \sin \theta + \Delta F_{11} \cos \theta)$$
$$\int_{r_0/r}^r B_{23}(\lambda \boldsymbol{x}) \lambda d\lambda = \frac{r - r_0}{2\pi r^2} (\Delta F_{22} \sin \theta + \Delta F_{21} \cos \theta),$$

therefore

$$\int_{0}^{1} B_{13}(\lambda \boldsymbol{x}) \lambda d\lambda = (\frac{1}{3\pi r_{0}} + \frac{r - r_{0}}{2\pi r^{2}}) (\Delta F_{12} \sin \theta + \Delta F_{11} \cos \theta)$$

$$\int_{0}^{1} B_{23}(\lambda \boldsymbol{x}) \lambda d\lambda = (\frac{1}{3\pi r_{0}} + \frac{r - r_{0}}{2\pi r^{2}}) (\Delta F_{22} \sin \theta + \Delta F_{21} \cos \theta).$$

Converting from polar coordinates to Cartesian coordinates,

$$\int_{0}^{1} B_{13}(\lambda \boldsymbol{x}) \lambda d\lambda = \left(\frac{1}{3\pi r_{0}r} + \frac{r-r_{0}}{2\pi r^{3}}\right) \left(\Delta F_{12}x_{2} + \Delta F_{11}x_{1}\right)$$
$$\int_{0}^{1} B_{23}(\lambda \boldsymbol{x}) \lambda d\lambda = \left(\frac{1}{3\pi r_{0}r} + \frac{r-r_{0}}{2\pi r^{3}}\right) \left(\Delta F_{22}x_{2} + \Delta F_{21}x_{1}\right),$$



Figure 25: A single g.disclination with a circular and an arbitrary path connecting x^- and x^+ in the simply-connected domain $(\Omega \setminus \Omega_c) \setminus S$.

and the solution for A^* is

$$\begin{split} A_{11}^{*} &= \begin{cases} (\frac{1}{2\pi r_{0}r} - \frac{1}{6\pi r_{0}^{2}})(-\Delta F_{12}x_{2}^{2} - \Delta F_{11}x_{1}x_{2}) & r < r_{0} \\ (\frac{1}{3\pi r_{0}r} + \frac{r-r_{0}}{2\pi r^{3}})(-\Delta F_{12}x_{2}^{2} - \Delta F_{11}x_{1}x_{2}) & r \geq r_{0} \end{cases} \\ A_{12}^{*} &= \begin{cases} (\frac{1}{2\pi r_{0}r} - \frac{1}{6\pi r_{0}^{2}})(\Delta F_{12}x_{2}x_{1} + \Delta F_{11}x_{1}^{2}) & r < r_{0} \\ (\frac{1}{3\pi r_{0}r} + \frac{r-r_{0}}{2\pi r^{3}})(\Delta F_{12}x_{2}x_{1} + \Delta F_{11}x_{1}^{2}) & r \geq r_{0} \end{cases} \\ A_{21}^{*} &= \begin{cases} (\frac{1}{2\pi r_{0}r} - \frac{1}{6\pi r_{0}^{2}})(-\Delta F_{22}x_{2}^{2} - \Delta F_{21}x_{1}x_{2}) & r < r_{0} \\ (\frac{1}{3\pi r_{0}r} + \frac{r-r_{0}}{2\pi r^{3}})(-\Delta F_{22}x_{2}^{2} - \Delta F_{21}x_{1}x_{2}) & r < r_{0} \end{cases} \\ A_{22}^{*} &= \begin{cases} (\frac{1}{2\pi r_{0}r} - \frac{1}{6\pi r_{0}^{2}})(-\Delta F_{22}x_{2}x_{1} + \Delta F_{21}x_{1}^{2}) & r < r_{0} \\ (\frac{1}{3\pi r_{0}r} + \frac{r-r_{0}}{6\pi r_{0}^{2}})(\Delta F_{22}x_{2}x_{1} + \Delta F_{21}x_{1}^{2}) & r < r_{0} \end{cases} \\ A_{22}^{*} &= \begin{cases} (\frac{1}{2\pi r_{0}r} - \frac{1}{6\pi r_{0}^{2}})(\Delta F_{22}x_{2}x_{1} + \Delta F_{21}x_{1}^{2}) & r < r_{0} \\ (\frac{1}{3\pi r_{0}r} + \frac{r-r_{0}}{2\pi r^{3}})(\Delta F_{22}x_{2}x_{1} + \Delta F_{21}x_{1}^{2}) & r < r_{0} \end{cases} \end{aligned}$$

C An auxiliary path-independence result in the dislocation-free case

Consider the configuration shown in Figure 25 with the whole domain denoted as Ω , the core as Ω_c , the cut-surface as S, and the cut-induced simply-connected domain as $(\Omega \setminus \Omega_c) \setminus S$. Given any g.disclination density localized in the core Ω_c (the patch enclosed by the red line in Figure 25) and the dislocation density $\boldsymbol{\alpha} = \mathbf{0}$, we can calculate S^* and A^* from the following

$$\begin{array}{c} \operatorname{curl} \boldsymbol{S}^* = \boldsymbol{\varPi} \\ \operatorname{div} \boldsymbol{S}^* = \boldsymbol{0} \end{array} & \text{ in } \boldsymbol{\varOmega} \\ \operatorname{curl} \boldsymbol{A}^* = \boldsymbol{S}^* : \boldsymbol{X} \\ \operatorname{div} \boldsymbol{A}^* = \boldsymbol{0} \end{array} & \text{ in } \boldsymbol{\varOmega} \end{array}$$

Given a fixed point \boldsymbol{x}_r , define $\tilde{\boldsymbol{T}}^p$ as

$$ilde{oldsymbol{T}}^p(oldsymbol{x};oldsymbol{x}_r):=\int_{oldsymbol{x}_r}^{oldsymbol{x}}oldsymbol{S}^*(oldsymbol{s})dolds+oldsymbol{A}^*(oldsymbol{x}),$$

where p is a path from \boldsymbol{x}_r to \boldsymbol{x} . Since $curl \boldsymbol{S}^* = \boldsymbol{0}$ outside the core Ω_c , $\tilde{\boldsymbol{T}}^p$ is path-independent in the simply-connected domain $(\Omega \setminus \Omega_c) \setminus S$; hence, we denote $\tilde{\boldsymbol{T}}^p(\boldsymbol{x}; \boldsymbol{x}_r)$ as $\tilde{\boldsymbol{T}}(\boldsymbol{x}; \boldsymbol{x}_r)$. Also, since \boldsymbol{A}^* is calculated from $curl \boldsymbol{A}^* = \boldsymbol{S}^* : \boldsymbol{X}$, we have

$$\operatorname{curl} \tilde{T}(\boldsymbol{x}; \boldsymbol{x}_r) = \operatorname{curl} \int_{\boldsymbol{x}_r}^{\boldsymbol{x}} S^*(\boldsymbol{s}) d\boldsymbol{s} + \operatorname{curl} A^*$$

 $\Rightarrow \operatorname{curl} \tilde{T}(\boldsymbol{x}; \boldsymbol{x}_r) = -\operatorname{grad} \left(\int_{\boldsymbol{x}_r}^{\boldsymbol{x}} S^*(\boldsymbol{s}) d\boldsymbol{s} \right) : \boldsymbol{X} + \operatorname{curl} A^*$
 $\Rightarrow \operatorname{curl} \tilde{T}(\boldsymbol{x}; \boldsymbol{x}_r) = -S^* : \boldsymbol{X} + \operatorname{curl} A^*$
 $\Rightarrow \operatorname{curl} \tilde{T}(\boldsymbol{x}; \boldsymbol{x}_r) = \boldsymbol{0}.$

Let C_1 be the anti-clockwise circular path from x^+ to x^- with radius R and C_2 be the clockwise outer path from x^- to x^+ shown in the Figure 25. Then $C = C_1 + C_2$ is the closed contour enclosing the blue shaded area. Take the integral of \tilde{T} along the closed contour C shown as in Figure 25,

$$\int_C \tilde{\boldsymbol{T}}(\boldsymbol{x};\boldsymbol{x}_r) d\boldsymbol{x} = \int_A \operatorname{curl} \tilde{\boldsymbol{T}}(\boldsymbol{x};\boldsymbol{x}_r) \boldsymbol{n} da,$$

where A is the patch enclosed by the loop C and n is the normal unit vector of the patch A. Therefore,

$$\int_{C} \tilde{T}(\boldsymbol{x}; \boldsymbol{x}_{r}) d\boldsymbol{x} = \boldsymbol{0}$$
$$\Rightarrow \int_{C_{1}} \tilde{T}(\boldsymbol{x}; \boldsymbol{x}_{r}) d\boldsymbol{x} + \int_{C_{2}} \tilde{T}(\boldsymbol{x}; \boldsymbol{x}_{r}) d\boldsymbol{x} = \boldsymbol{0}$$
$$\Rightarrow \int_{-C_{1}} \tilde{T}(\boldsymbol{x}; \boldsymbol{x}_{r}) d\boldsymbol{x} = \int_{C_{2}} \tilde{T}(\boldsymbol{x}; \boldsymbol{x}_{r}) d\boldsymbol{x}.$$

Namely, the integration of $\tilde{T}(\boldsymbol{x};\boldsymbol{x}_r)$ from \boldsymbol{x}^- to \boldsymbol{x}^+ along any path can be calculated as the integration of $\tilde{T}(\boldsymbol{x};\boldsymbol{x}_r)$ along a circular path from \boldsymbol{x}^- to \boldsymbol{x}^+ .

D Calculations for Burgers vector of a g.disclination dipole

Given the configuration in Section 8, and for in-plane variations of fields, Π^+ and Π^- only have non-zero component Π^+_{ij3} and Π^-_{ij3} , which are given as follows:

$$\Pi_{ij3}^{+} = \begin{cases} \frac{\Delta_{ij}^{F}}{\pi r_{0}} \left(\frac{1}{r_{+}} - \frac{1}{r_{0}} \right) & r_{+} < r_{0} \\ 0 & r_{+} \ge r_{0}, \end{cases}$$

and

$$\Pi_{ij3}^{-} = \begin{cases} -\frac{\Delta_{ij}^{F}}{\pi r_{0}} \left(\frac{1}{r_{-}} - \frac{1}{r_{0}}\right) & r_{-} < r_{0} \\ 0 & r_{-} \ge r_{0}, \end{cases}$$

where $r_+ = \sqrt{x_1^2 + x_2^2}$ and $r_- = \sqrt{(x_1 - d)^2 + x_2^2}$. As given in Appendix A, we obtain S^{*+} as

$$S_{ij1}^{*+} = \begin{cases} \frac{\Delta_{ij}^{F}}{2\pi} \left(-\frac{x_{2}}{r^{2}}\right) & r_{+} > r_{0} \\ \frac{-x_{2}\Delta_{ij}^{F}}{\pi r_{+}^{2}r_{0}} \left(r_{+} - \frac{r_{+}^{2}}{2r_{0}}\right) & r_{+} \le r_{0} \end{cases}$$
$$S_{ij2}^{*+} = \begin{cases} \frac{\Delta_{ij}^{F}}{2\pi} \left(\frac{x_{1}}{r_{+}^{2}}\right) & r_{+} > r_{0} \\ \frac{x_{1}\Delta_{ij}^{F}}{\pi r_{+}^{2}r_{0}} \left(r_{+} - \frac{r_{+}^{2}}{2r_{0}}\right) & r_{+} \le r_{0}, \end{cases}$$

and S^{*-} as

$$S_{ij1}^{*-} = \begin{cases} \frac{\Delta_{ij}^{F}}{2\pi} \left(\frac{x_{2}}{r^{2}}\right) & r_{-} > r_{0} \\ \frac{x_{2}\Delta_{ij}^{F}}{\pi r_{-}^{2}r_{0}} \left(r_{-} - \frac{r_{-}^{2}}{2r_{0}}\right) & r_{-} \le r_{0} \end{cases}$$
$$S_{ij2}^{*-} = \begin{cases} \frac{\Delta_{ij}^{F}}{2\pi} \left(-\frac{x_{1}-d}{r_{-}^{2}}\right) & r_{-} > r_{0} \\ \frac{-(x_{1}-d)\Delta_{ij}^{F}}{\pi r_{-}^{2}r_{0}} \left(r_{-} - \frac{r_{-}^{2}}{2r_{0}}\right) & r_{-} \le r_{0} \end{cases}$$

Also, following the same arguments as in Appendix B, we obtain for A^{*+} and A^{*-} :

$$\begin{split} A_{11}^{*+} &= \begin{cases} C_1^+ \left(-\Delta_{12}^F x_2^2 - \Delta_{11}^F x_1 x_2 \right) & r_+ < r_0 \\ C_2^+ \left(-\Delta_{12}^F x_2^2 - \Delta_{11}^F x_1 x_2 \right) & r_+ \ge r_0 \end{cases} \\ A_{12}^{*+} &= \begin{cases} C_1^+ \left(\Delta_{12}^F x_2 x_1 + \Delta_{11}^F x_1^2 \right) & r_+ < r_0 \\ C_2^+ \left(\Delta_{12}^F x_2 x_1 + \Delta_{11}^F x_1^2 \right) & r_+ \ge r_0 \end{cases} \\ A_{21}^{*+} &= \begin{cases} C_1^+ \left(-\Delta_{22}^F x_2^2 - \Delta_{21}^F x_1 x_2 \right) & r_+ < r_0 \\ C_2^+ \left(-\Delta_{22}^F x_2^2 - \Delta_{21}^F x_1 x_2 \right) & r_+ \ge r_0 \end{cases} \\ A_{22}^{*+} &= \begin{cases} C_1^+ \left(\Delta_{22}^F x_2 x_1 + \Delta_{21}^F x_1^2 \right) & r_+ < r_0 \\ C_2^+ \left(\Delta_{22}^F x_2 x_1 + \Delta_{21}^F x_1^2 \right) & r_+ \ge r_0 \end{cases} \\ A_{11}^{*-} &= \begin{cases} C_1^- \left(\Delta_{12}^F x_2^2 + \Delta_{11}^F (x_1 - d) x_2 \right) & r_- < r_0 \\ C_2^- \left(\Delta_{12}^F x_2^2 + \Delta_{11}^F (x_1 - d) x_2 \right) & r_- \ge r_0 \end{cases} \\ A_{12}^{*-} &= \begin{cases} C_1^- \left(\Delta_{22}^F x_2^2 + \Delta_{21}^F (x_1 - d) x_2 \right) & r_- < r_0 \\ C_2^- \left(-\Delta_{12}^F x_2 (x_1 - d) - \Delta_{111}^F (x_1 - d)^2 \right) & r_- \ge r_0 \end{cases} \\ A_{21}^{*-} &= \begin{cases} C_1^- \left(\Delta_{22}^F x_2^2 + \Delta_{21}^F (x_1 - d) x_2 \right) & r_- < r_0 \\ C_2^- \left(-\Delta_{12}^F x_2 (x_1 - d) - \Delta_{111}^F (x_1 - d)^2 \right) & r_- \ge r_0 \end{cases} \\ A_{21}^{*-} &= \begin{cases} C_1^- \left(-\Delta_{22}^F x_2 (x_1 - d) - \Delta_{21}^F (x_1 - d) x_2 \right) & r_- < r_0 \\ C_2^- \left(-\Delta_{22}^F x_2 (x_1 - d) - \Delta_{21}^F (x_1 - d) x_2 \right) & r_- \ge r_0 \end{cases} \\ A_{22}^{*-} &= \begin{cases} C_1^- \left(-\Delta_{22}^F x_2 (x_1 - d) - \Delta_{21}^F (x_1 - d) x_2 \right) & r_- < r_0 \\ C_2^- \left(-\Delta_{22}^F x_2 (x_1 - d) - \Delta_{21}^F (x_1 - d) x_2 \right) & r_- \ge r_0 \end{cases} \end{aligned}$$

where $C_1^+ = \frac{1}{2\pi r_0 r_+} - \frac{1}{6\pi r_0^2}$, $C_2^+ = \frac{1}{3\pi r_0 r_+} + \frac{r_+ - r_0}{2\pi r_+^3}$, $C_1^- = \frac{1}{2\pi r_0 r_-} - \frac{1}{6\pi r_0^2}$ and $C_2^- = \frac{1}{3\pi r_0 r_-} + \frac{r_- - r_0}{2\pi r_-^3}$. Given \boldsymbol{x}_+ as $(-x_0, 0)$, \boldsymbol{x}_- as $(d - x_0, 0)$ and two paths p^+ and p^- as in Figure 24, following the same notation as in Section 8, we have

$$\begin{split} \oint_{p^+} \left\{ \int_{\boldsymbol{x}_-}^{\boldsymbol{x}} \boldsymbol{S}^{*+}(\boldsymbol{s}) d\boldsymbol{s} + \boldsymbol{A}^{*+}(\boldsymbol{x}) \right\} d\boldsymbol{x} &=: \boldsymbol{I}^+ \\ \oint_{p^-} \left\{ \int_{\boldsymbol{x}_-}^{\boldsymbol{x}} \boldsymbol{S}^{*-}(\boldsymbol{s}) d\boldsymbol{s} + \boldsymbol{A}^{*-}(\boldsymbol{x}) \right\} d\boldsymbol{x} &=: \boldsymbol{I}^-. \end{split}$$

After substituting S^{*+} , S^{*-} , A^{*+} , and A^{*-} , applying polar coordinates and some algebraic calculation, we obtain

$$I_1^+ = x_0 \Delta_{11}^F$$

$$I_2^+ = x_0 \Delta_{21}^F$$

$$I_1^- = -(x_0 - d) \Delta_{11}^F$$

$$I_2^- = -(x_0 - d) \Delta_{21}^F.$$

References

- [Ach01] Amit Acharya, A model of crystal plasticity based on the theory of continuously distributed dislocations, Journal of the Mechanics and Physics of Solids **49** (2001), no. 4, 761–784.
- [AD15] Vaibhav Agrawal and Kaushik Dayal, A dynamic phase-field model for structural transformations and twinning: Regularized interfaces with transparent prescription of complex kinetics and nucleation. Part I: Formulation and one-dimensional characterization, Journal of the Mechanics and Physics of Solids 85 (2015), 270–290.
- [AF12] Amit Acharya and Claude Fressengeas, Coupled phase transformations and plasticity as a field theory of deformation incompatibility, International Journal of Fracture 174 (2012), no. 1, 87–94.
- [AF15] _____, Continuum mechanics of the interaction of phase boundaries and dislocations in solids, Differential Geometry and Continuum Mechanics, Springer Proceedings in Mathematics and Statistics; Ed: G. Q. Chen, M. Grinfeld, R. J. Knops 137 (2015), 125–168.
- [AZH08] S. Akarapu, H. Zbib, and J. P. Hirth, Modeling and analysis of disconnections in tilt walls, Scripta Materialia 59 (2008), no. 3, 265–267.
- [BAC05] Robert W. Balluffi, Samuel M. Allen, and W. Craig Carter, *Kinetics of materials*, John Wiley & Sons, 2005.
- [BB56] R. Bullough and B. A. Bilby, Continuous distributions of dislocations: Surface dislocations and the crystallography of martensitic transformations, Proceedings of the Physical Society. Section B 69 (1956), no. 12, 1276.
- [BBS55] B. A. Bilby, R. Bullough, and E. Smith, Continuous Distributions of Dislocations: A New Application of the Methods of Non-Riemannian Geometry, Proceedings of the Royal Society of London A: Mathematical, Physical and Engineering Sciences 231 (1955), no. 1185, 263–273.
- [Cah82] J.W. Cahn, Transitions and phase equilibria among grain boundary structures, Le Journal de Physique Colloques 43 (1982), no. C6, C6–199.
- [Cas04] James Casey, On volterra dislocations of finitely deforming continua, Mathematics and mechanics of solids 9 (2004), no. 5, 473–492.
- [che] Chemdoodle, http://www.chemdoodle.com/.
- [CMB06] John D. Clayton, David L. McDowell, and Douglas J. Bammann, Modeling dislocations and disclinations with finite micropolar elastoplasticity, International Journal of Plasticity 22 (2006), no. 2, 210–256.
- [CMS06] John W. Cahn, Yuri Mishin, and Akira Suzuki, Coupling grain boundary motion to shear deformation, Acta Materialia 54 (2006), no. 19, 4953–4975.
- [DeW69] R. DeWit, *Linear theory of static disclinations*, Journal of Research of the National Bureau of Standards Section A Physics and Chemistry A 73 (1969), no. 5, 528–&.

- [DeW71] _____, Relation between dislocations and disclinations, Journal of Applied Physics 42 (1971), 3304–3308.
- [DeW72] R. DeWit, *Partial disclinations*, Journal of Physics C: Solid State Physics 5 (1972), no. 5, 529.
- [DeW73a] R. DeWit, Theory of disclinations. II- continuous and discrete disclinations in anisotropic elasticity, Journal of Research 77 (1973), 49–100.
- [DeW73b] R. DeWit, Theory of disclinations: IV. Straight disclinations, J. Res. Natl Bureau Standards Sect. A, Phys. Chem. A 77 (1973), 607–658.
- [DSSS98] Liang Dong, Jurgen Schnitker, Richard W. Smith, and David J. Srolovitz, Stress relaxation and misfit dislocation nucleation in the growth of misfitting films: A molecular dynamics simulation study, Journal of Applied Physics 83 (1998), no. 1, 217–227.
- [DXS13] Shuyang Dai, Yang Xiang, and David J. Srolovitz, *Structure and energy of (111) low-angle twist boundaries in Al, Cu and Ni*, Acta Materialia **61** (2013), no. 4, 1327–1337.
- [DZ11] S. Derezin and L. Zubov, Disclinations in nonlinear elasticity, ZAMM-Journal of Applied Mathematics and Mechanics/Zeitschrift f
 ür Angewandte Mathematik und Mechanik 91 (2011), no. 6, 433–442.
- [Ede85] Dominic G. B. Edelen, Applied Exterior Calculus, Wiley-Interscience, Wiley, New York, 1985.
- [EES⁺09] Matt Elsey, Selim Esedog, Peter Smereka, et al., Diffusion generated motion for grain growth in two and three dimensions, Journal of Computational Physics 228 (2009), no. 21, 8015–8033.
- [EL88] Dominic G. B. Edelen and Dimitris C. Lagoudas, Dispersion relations for the linearized field equations of dislocation dynamics, International journal of engineering science 26 (1988), no. 8, 837–846.
- [Fra50] F. C. Frank, The resultant content of dislocations in an arbitrary intercrystalline boundary, Symposium on The Plastic Deformation of Crystalline Solids, Mellon Institute, Pittsburgh,(NAVEXOS-P-834), vol. 150, 1950.
- [Fra53] F. C. Frank, *Martensite*, Acta Metallurgica 1 (1953), no. 1, 15 21.
- [FTC11] C. Fressengeas, V. Taupin, and L. Capolungo, An elasto-plastic theory of dislocation and disclination fields, International Journal of Solids and Structures 48 (2011), no. 25, 3499–3509.
- [FTC14] _____, Continuous modeling of the structure of symmetric tilt boundaries, International Journal of Solids and Structures **51** (2014), no. 6, 1434–1441.
- [HH98] J. P. Hirth and R. G. Hoagland, *Extended dislocation barriers in tilt boundaries in fcc crystals*, Material Research Innovations **1** (1998), no. 4, 235–237.
- [HHM01] E. A. Holm, G. N. Hassold, and M. A. Miodownik, On misorientation distribution evolution during anisotropic grain growth, Acta Materialia **49** (2001), no. 15, 2981–2991.

- [HP96] J. P. Hirth and R. C. Pond, Steps, dislocations and disconnections as interface defects relating to structure and phase transformations, Acta Materialia 44 (1996), no. 12, 4749–4763.
- [HP11] _____, Compatibility and accommodation in displacive phase transformations, Progress in Materials Science **56** (2011), no. 6, 586–636.
- [HPH09] J. M. Howe, R. C. Pond, and J. P. Hirth, The role of disconnections in phase transformations, Progress in Materials Science 54 (2009), no. 6, 792–838.
- [HPH⁺13] J. P. Hirth, R. C. Pond, R. G. Hoagland, X.-Y. Liu, and J. Wang, Interface defects, reference spaces and the Frank-Bilby equation, Progress in Materials Science 58 (2013), no. 5, 749–823.
- [HPL06] J. P. Hirth, R. C. Pond, and J. Lothe, Disconnections in tilt walls, Acta Materialia 54 (2006), no. 16, 4237–4245.
- [HPL07] _____, Spacing defects and disconnections in grain boundaries, Acta Materialia 55 (2007), no. 16, 5428–5437.
- [KF08] Maurice Kleman and Jacques Friedel, Disclinations, dislocations, and continuous defects: A reappraisal, Reviews of Modern Physics 80 (2008), no. 1, 61.
- [KLT06] David Kinderlehrer, Irene Livshits, and Shlomo Ta'Asan, A variational approach to modeling and simulation of grain growth, SIAM Journal on Scientific Computing 28 (2006), no. 5, 1694–1715.
- [KMS15] Raz Kupferman, Michael Moshe, and Jake P. Solomon, Metric description of singular defects in isotropic materials, Archive for Rational Mechanics and Analysis 216 (2015), no. 3, 1009–1047.
- [Krö81] Ekkehart Kröner, Continuum theory of defects, Physics of Defects (Jean-Paul Poirier Roger Balian, Maurice Kléman, ed.), vol. 35, North-Holland, Amsterdam, 1981, pp. 217– 315.
- [KRR06] Chang-Soo Kim, Anthony D. Rollett, and Gregory S. Rohrer, Grain boundary planes: New dimensions in the grain boundary character distribution, Scripta Materialia 54 (2006), no. 6, 1005–1009.
- [Li72] J. C. M. Li, Disclination model of high angle grain boundaries, Surface Science **31** (1972), 12–26.
- [Mul56] William W. Mullins, *Two-Dimensional Motion of Idealized Grain Boundaries*, Journal of Applied Physics **27** (1956), no. 8, 900–904.
- [Nab85] F. R. N. Nabarro, The development of the idea of a crystal dislocation, University of Tokyo Press, Tokyo, 1985.
- [Nab87] _____, Theory of Crystal Dislocations, Dover Pubns, 1987.
- [Naz13] Ayrat A. Nazarov, Disclinations in bulk nanostructured materials: their origin, relaxation and role in material properties, Advances in Natural Sciences. Nanoscience and Nanotechnology 4 (2013), no. 3, 033002–1.

- [NSB00] A. A. Nazarov, O. A. Shenderova, and D. W. Brenner, On the disclination-structural unit model of grain boundaries, Materials Science and Engineering: A 281 (2000), no. 1, 148–155.
- [PAD15] Hossein Pourmatin, Amit Acharya, and Kaushik Dayal, A fundamental improvement to Ericksen-Leslie kinematics, Quarterly of Applied Mathematics LXXXIII (2015), no. 3, 435–466.
- [PL13] Marcel Porta and Turab Lookman, Heterogeneity and phase transformation in materials: Energy minimization, iterative methods and geometric nonlinearity, Acta Materialia 61 (2013), no. 14, 5311–5340.
- [RG17] Ayan Roychowdhury and Anurag Guptan, Non-metric connection and metric anomalies in materially uniform elastic solids, Journal of Elasticity **126** (2017), no. 14, 1–26.
- [RK09] Alexey E. Romanov and Anna L. Kolesnikova, *Application of disclination concept to* solid structures, Progress in Materials Science **54** (2009), no. 6, 740–769.
- [Roh10] Gregory S. Rohrer, Introduction to Grains, Phases, and Interfacesan Interpretation of Microstructure, Trans. AIME, 1948, vol. 175, pp. 15–51, by C.S. Smith, Metallurgical and Materials Transactions A 41 (2010), no. 5, 1063–1100.
- [Roh11] _____, Grain boundary energy anisotropy: a review, Journal of Materials Science 46 (2011), no. 18, 5881–5895.
- [RS50] W. T. Read and W. Shockley, Dislocation models of crystal grain boundaries, Physical Review 78 (1950), no. 3, 275.
- [RV92] A. E. Romanov and V. I. Vladimirov, Disclinations in Crystalline Solids, Dislocation in Solids 9 (1992), 191–402.
- [SB95] Adrian P. Sutton and Robert W. Balluffi, *Interfaces in crystalline materials*, Clarendon Press, 1995.
- [SEIDRR04] David M. Saylor, Bassem S. E. I. Dasher, Anthony D. Rollett, and Gregory S. Rohrer, Distribution of grain boundaries in aluminum as a function of five macroscopic parameters, Acta Materialia 52 (2004), no. 12, 3649–3655.
- [SV83a] A. P. Sutton and V. Vitek, On the structure of tilt grain boundaries in cubic metals I. Symmetrical tilt boundaries, Philosophical Transactions of the Royal Society of London A: Mathematical, Physical and Engineering Sciences **309** (1983), no. 1506, 1–36.
- [SV83b] _____, On the structure of tilt grain boundaries in cubic metals II. Asymmetrical tilt boundaries, Philosophical Transactions of the Royal Society of London A: Mathematical, Physical and Engineering Sciences **309** (1983), no. 1506, 37–54.
- [VAKD14] A. J. Vattré, N. Abdolrahim, K. Kolluri, and M. J. Demkowicz, Computational design of patterned interfaces using reduced order models, Scientific Reports 4 (2014).
- [VD13] A. J. Vattré and M. J. Demkowicz, Determining the Burgers vectors and elastic strain energies of interface dislocation arrays using anisotropic elasticity theory, Acta Materialia 61 (2013), no. 14, 5172–5187.

- [VD15] _____, Partitioning of elastic distortions at a semicoherent heterophase interface between anisotropic crystals, Acta Materialia 82 (2015), 234–243.
- [Wil67] John R. Willis, Second-order effects of dislocations in anisotropic crystals, International Journal of Engineering Science 5 (1967), no. 2, 171–190.
- [YG12] Arash Yavari and Alain Goriely, *Riemann–Cartan geometry of nonlinear disclination mechanics*, Mathematics and Mechanics of Solids (2012), 1081286511436137.
- [ZAP16] Chiqun Zhang, Amit Acharya, and Saurabh Puri, Finite element approximation of the fields of bulk and interfacial line defects. Unpublished results for manuscript in preparation (2016).
- [Zub97] Leonid M. Zubov, Nonlinear theory of dislocations and disclinations in elastic bodies, vol. 47, Springer Science & Business Media, 1997.